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# Discrete-Trap Effects on 3-D NAND Variability – Part II: Random Telegraph Noise

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**ABSTRACT** In Part II of this article we discuss the impact of a discrete treatment of traps on 3-D NAND Flash random telegraph noise (RTN). A higher RTN results when discrete traps are taken into account, that can only be explained by a stronger influence of the discrete charged traps on the current conduction, leading to more percolation. The effects are then investigated as a function of the cell parameters, showing that a continuous model for traps cannot reproduce the correct dependence.

**INDEX TERMS** 3-D NAND Flash memories, variability, random telegraph noise, discrete traps.

## I. INTRODUCTION

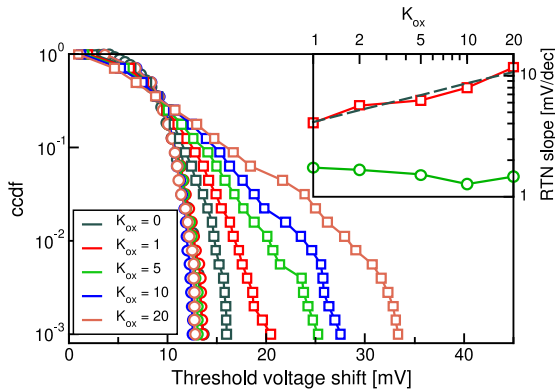
In Part I of this work we discussed the improvement in device modeling that follows the adoption of a discrete treatment of the traps in the polysilicon channel of 3-D NAND cells, showing that models relying on a continuous trap (CT) approach overestimate  $V_T$ , as a discrete-trap (DT) treatment allows for additional current percolation and fewer occupied traps. Here, we extend such analysis to random telegraph noise (RTN), *i.e.*,  $V_T$  fluctuations following the capture of an electron by a trap. RTN is an important reliability issue in 3-D NAND cells [1], [2], [3], affecting the  $V_T$  distribution width [4], [5] while acting as a process monitor [6], [7]. Indeed, RTN has been already investigated by several authors in a CT [8], [9], [10], [11], [12], [13], [14], [15], [16] and DT [17], [18] approach, but a thorough comparison between CT and DT results is still missing. As with Part I, preliminary results have been presented in [19]; here we expand the physical analysis and discuss the dependence on device parameters and temperature. Results of this work make clear that a CT model cannot reproduce the correct dependence on device parameters provided by a full DT approach.

## II. SIMULATION RESULTS

We refer the reader to Part I [20] for details and numerical values of the device parameters and just recall here the main

features of our simulations: The structure is a 3-D NAND string consisting of a central cell plus two half-cells per side. Memory cells have a Macaroni structure [21], made of an inner filler oxide, a thin layer of polysilicon channel with 10 nm thickness, an oxide/nitride/oxide stack, and a metal wordline with length  $L = 35$  nm (see Fig. 1 in Part I). Polysilicon is described by random grains with average size  $D_g = 30$  nm and traps located at their grain boundaries (GBs), having an energy distribution derived from [22]. A polysilicon/oxide interface state density has been also considered.

3-D Monte Carlo simulations were performed using Sentaurus Process [23] and Sentaurus Device [24] for processing the typical cell structure and Garand VE [25] for studying the statistical behavior of the cells. For the simulation of RTN we follow a standard procedure [26] and begin with performing a drain current vs. wordline voltage  $I - V$  simulation up to the threshold level of 10 nA. Then, one additional trap is filled with an electron and a second  $I - V$  curve is repeated, leading to a higher  $V_T$ . The RTN fluctuation  $\Delta V_T$  is the difference value. This procedure is repeated with random grain number and positions, randomly locating the RTN traps either at the polysilicon/oxide interfaces or at GBs, depending on the relative densities. It is important to note that the RTN



**FIGURE 1.**  $\Delta V_T$  ccdf for different values of the polysilicon/oxide interface trap density. Inset shows the RTN slope as a function of  $K_{ox}$ , with the dashed line highlighting a power law with exponent 0.32. In both graphs: Circles = CTs, squares = DTs.

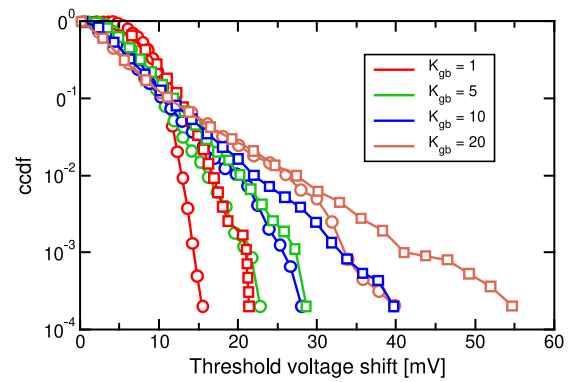
trap is always treated as discrete, irrespective of the model adopted for all the other ones. Note also that this procedure amounts to computing the single-trap RTN distribution, that can however be easily linked to the real multi-trap one [27].

### A. POLYSILICON/OXIDE INTERFACE TRAPS

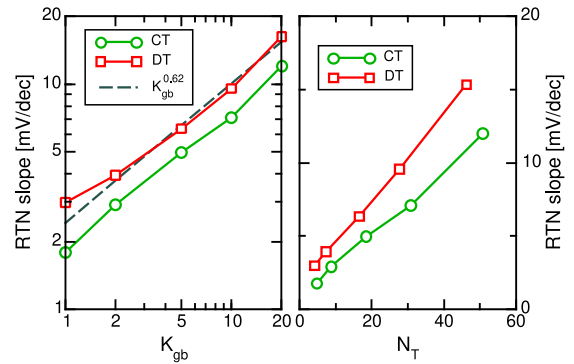
Fig. 1 shows the  $\Delta V_T$  complementary cumulative distribution function (ccdf) when the polysilicon/oxide interface state density is multiplied by a factor  $K_{ox}$ . Note that such increase has no effect on the CT  $\Delta V_T$  distributions (circles), that present vanishingly small exponential tails with inverse slope  $\lambda$  of less than 2 mV/decade, but results in clear exponential distributions with  $\lambda$  increasing with  $K_{ox}$  whenever DTs are accounted for. In fact, in the CT case the trap distribution is uniform along the device, and the only percolation source is the GB traps, resulting in  $\lambda$  being independent of  $K_{ox}$ . In the DT case, instead, the Coulomb peaks at the filled trap sites generate a percolative current flow, by analogy with the influence of DTs on  $V_T$  already discussed in Part I. The magnitude of such effect is displayed in the inset of Fig. 1:  $\lambda$  in the DT case ( $\lambda_{DT}$ ) changes from about 4 to 12 mV/decade in the investigated range, and follows a power-law with exponent 0.32. For  $K_{ox} = 20$ ,  $\lambda_{DT}$  is about a factor of 8 larger than its CT counterpart ( $\lambda_{CT}$ ).

### B. GRAIN BOUNDARY TRAPS

To study the GB trap density dependence, we first removed the oxide traps from the following simulations, in order not to overshadow the low-density data, where they become dominant.  $\Delta V_T$  ccdfs are shown in Fig. 2 when the GB trap density is multiplied by a factor  $K_{gb}$ : both distributions are now affected by  $K_{gb}$ , although larger  $\Delta V_T$  values are reached in the DT case. The  $\lambda$  dependence on  $K_{gb}$  is plotted in Fig. 3 (left): both models return a power-law with exponent of about 0.62 (dashed line), with  $\lambda_{DT}$  remaining higher by about 35% with respect to  $\lambda_{CT}$ . In terms of trap density, this means that an increase by a factor of 58% is required to a CT approach in order to obtain the correct slope. For  $K_{gb} = 1$ ,



**FIGURE 2.**  $\Delta V_T$  ccdf for different values of the GB trap density with  $K_{ox} = 0$ . Circles = CTs, squares = DTs.



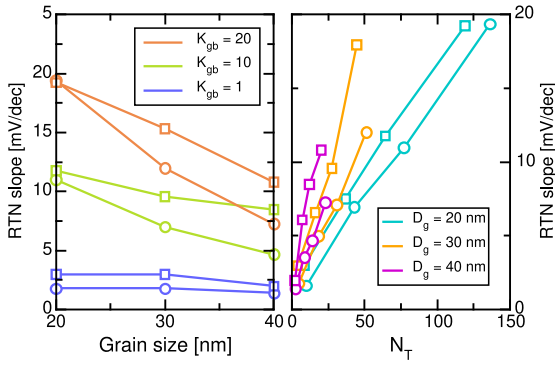
**FIGURE 3.** Left:  $\lambda$  as a function of the GB trap density. The dashed line shows a power law with exponent 0.62. Right:  $\lambda$  as a function of the number of filled traps in the cell channel region. In both graphs: Circles = CTs, squares = DTs.

our data show a deviation from the power law. However, the exponential tail of the  $\Delta V_T$  distribution cannot be neatly distinguished here, and a larger number of simulations would be required for a reliable extraction of  $\lambda$ .

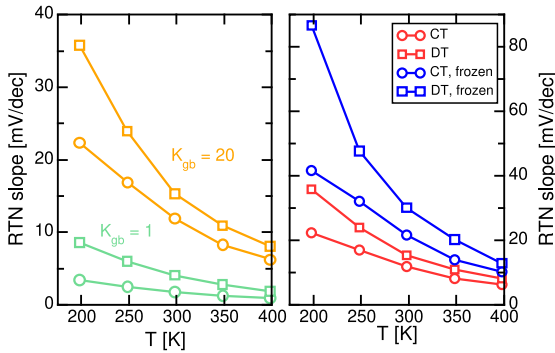
Fig. 3 (right) shows  $\lambda$  versus the average number of occupied traps  $N_T$ . As a difference from  $V_T$  in Part I, no unique behavior is observed, and  $\lambda_{DT}$  is higher by about 50% with respect to  $\lambda_{CT}$ , confirming the stronger percolation through discrete charged traps. In the CT case, instead, the RTN is due to the randomness in the GB configuration [11].

### C. GRAIN SIZE

Fig. 4 (left) shows the dependence of  $\lambda$  on  $D_g$  for different values of  $K_{gb}$ . The increase in  $\lambda$  with smaller grains has been reported in [15] and can be understood recalling (see Part I) that the GB surface trap density is independent of  $D_g$ , meaning that decreasing  $D_g$  we obtain more grains and GBs, hence more traps. More important, results show that the difference between  $\lambda_{CT}$  and  $\lambda_{DT}$  becomes less relevant for smaller  $D_g$ . To clarify this point, we can look at data as a function of  $N_T$  (right), showing that the impact of traps on  $\lambda$  decreases for smaller grains, where we have more GBs, hence a more even distribution of traps and weaker percolation, *i.e.*, smaller  $\lambda$  and smaller differences between



**FIGURE 4.** RTN slope as a function of the grain size (left) and  $N_T$  (right). Circles = CTs, squares = DTs.



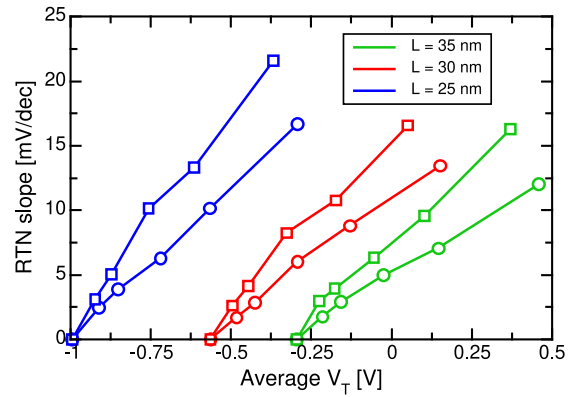
**FIGURE 5.** Left:  $\lambda$  as a function of temperature for  $K_{gb} = 1$  and 20. Right: Impact of traps equilibrium condition on the temperature dependence of  $\lambda$  for  $K_{gb} = 20$ . In both graphs: Circles = CTs, Squares = DTs.

$\lambda_{CT}$  and  $\lambda_{DT}$ . Finally, when data are plotted as a function of  $K_{gb}$  (not shown), a behavior similar to Fig. 3 arises, with a power-law exponent changing from 0.56 ( $D_g = 40$  nm) to 0.71 ( $D_g = 20$  nm).

#### D. TEMPERATURE

Fig. 5 (left) shows  $\lambda$  as a function of temperature  $T$  for the limiting cases of  $K_{gb} = 1$  and 20. The increase of  $\lambda$  at low  $T$  has been already discussed in [11], but the main result is the magnitude of the difference between  $\lambda_{CT}$  and  $\lambda_{DT}$ , that grows up to about 100% for  $K_{gb} = 1$  and 60% for  $K_{gb} = 20$ , at the lowest considered  $T$ . This result makes the adoption of a DT approach even more compelling in low-temperature applications.

As  $T$  is lowered, however, the role played by the time constants of capture/emission to/from traps should be considered. In fact, as an RTN trap is filled by an electron, the occupation probability of nearby traps lowers slightly, as the conduction band is locally raised. Traps will then reach the new equilibrium condition within a few capture/emission time constants. However, at low  $T$  such time constants increase [12], and it becomes harder for traps to reach the new equilibrium condition. A full treatment of this phenomenon would require a time-dependent Monte Carlo, but the magnitude of the effect can be grasped by comparing



**FIGURE 6.** RTN slope as a function of the average  $V_T$  for different values of  $L$ . Circles = CTs, squares = DTs.

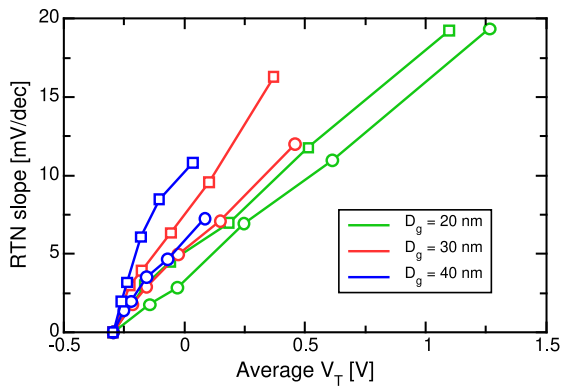
two approaches: one in which all trap occupations reach the equilibrium condition after filling of the RTN trap (our default case), and one where trap occupations are “frozen” to the pre-capture condition: As  $T$  is reduced, we could envisage a transition from the former to the latter regime. Data for  $K_{gb} = 1$  (not shown) feature no difference in the CT case and an increase by 17% in  $\lambda_{DT}$  at 198 K; data for  $K_{gb} = 20$  are reported in Fig. 5 (right), showing a very significant difference, with slopes increasing by about 60% at high  $T$  and by more than a factor of 2 in  $\lambda_{DT}$  at 198 K (where the increase in  $\lambda_{CT}$  is about 86%).

#### III. DISCUSSION

It remains now to be seen whether DT effects can be reproduced by a CT model via a change in the trap density or not. In such case, a CT approach could still be employed, although with the wrong value for the trap density; otherwise, a DT approach would become necessary to reproduce the correct dependences. The latter case is obviously the one for the interface trap density, where the RTN slopes are very different (see inset of Fig. 1). For the  $K_{gb}$  dependence, we reported  $\lambda$  as a function of the average  $V_T$  (taken from Part I) in Fig. 6, for all values of  $K_{gb}$ , extending the study to different wordline lengths  $L$ . So, each curve represents the possible values of  $V_T$  and  $\lambda$  that can be achieved with a given model when the trap density is changed. As CT and DT curves have different slopes, a DT approach is needed to capture the correct dependences.

As for the dependence on  $L$ , results for crystalline planar [28] and nanowire devices [26] already showed that  $\lambda$  increases in shorter channels, here by a factor of about 35 – 40% (similar for  $\lambda_{CT}$  and  $\lambda_{DT}$ ) as  $L$  is shrunk from 35 to 25 nm. Such trends are the results of different phenomena, including an increased impact of charged traps in shorter channels and a reduction in their number. Moreover, short-channel effects also play a contribution, making it harder for the gate to control the channel conduction and offset the impact of the RTN trap.

Finally, Fig. 7 shows results similar to Fig. 6, but for different values of  $D_g$  (see Fig. 4). In this case as well,



**FIGURE 7.** RTN slope as a function of the average  $V_T$  for different values of  $D_g$ . Circles = CTs, squares = DTs.

no overlap can be seen between the CT and DT curves, strengthening the need for a treatment of the discrete nature of traps.

#### IV. CONCLUSION

We have studied the impact of a DT approach on RTN in 3-D NAND with polysilicon channel. DT simulations lead to larger RTN and stronger dependence on the trap density, which is explained in terms of a stronger percolation through discretely charged traps. The dependence on the cell parameters shows that a model based on CTs cannot reproduce the correct dependences predicted by a DT-based one.

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