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1 Insights into the printing parameters and characterization of 2 thermoplastic polyurethane soft triply periodic minimal surface and 3 honeycomb lattices for broadening material extrusion applicability

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11 12 Abstract

13 Lattice structures with triply periodic minimal surfaces (TPMS) built using flexible materials are
14 soft porous solids applicable in various fields, including biomedicine and tissue engineering. Such
15 structures are also relevant for material extrusion additive manufacturing (MEAM), whose wide
16 diffusion is pivotal to fostering their spread. Although design approaches are available to exploit the
17 potential of soft TPMS, there are still manufacturing constraints that lead to practical limits on the
18 shape and size of the structures that can be produced due to the complexities related to printing
19 flexible materials. Besides, the computational models investigating the effect of cell type, the
20 surface-to-volume fraction, and the combination of different periodic surfaces (i.e., graded or
21 hybrid) on the mechanical behavior of these lattices are design aspects still debated. Here, the
22 capabilities of MEAM to produce tailored soft lattice structures are explored by combining a design
23 tool, numerical analyses, and mechanical testing using thermoplastic polyurethane (TPU) as
24 feedstock material. The study addresses design issues, delves into optimum printing parameters, and
25 analyzes a set of numerical parameters, which can be used for designing specific structures with
26 tunable mechanical behavior, useful for healthcare and bioengineering. The printing parameters of
27 three lattices, i.e., schwartz-P, gyroid, and honeycomb, with unit cell sizes spanning from 3 to 12
28 mm were studied. Their mechanical behavior was investigated using FEM simulations and
29 mechanical testing. Lastly, the printability of graded and hybrid lattices with enhanced bearing-load
30 capabilities have been demonstrated. Altogether, our findings addressed multiple challenges
31 associated with developing soft lattice scaffolds with MEAM that can be used to fabricate
32 innovative-engineered materials with tunable properties.

33
34 **Keywords:** fused filament fabrication (FFF); design for additive manufacturing; thermoplastic
35 polyurethane (TPU); finite element analysis (FEM); lattice structures; triply periodic minimal
36 surfaces (TPMS).

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41 **1. Introduction**

42 The field of engineering design lattice structures is continuously growing in different sectors such
43 as biomedical, tissue engineering, orthopedics, aerospace, and automotive [1-9]. Nature has played
44 a crucial role for scientists to understand and explore such structures [10-12]; just think of the
45 geometry, shape, and mechanical properties of shells, shark teeth, honeycomb, cancellous bone, and
46 marine sponges. Indeed, nature has found a fascinating way to use geometric lattice design
47 principles, allowing structures (or molecules) to hierarchically self-assemble from nano- to meso-
48 scale level, thus leading to exceptional properties [12]. Among the above-mentioned structures,
49 lattices with periodic minimal surfaces and negative Gaussian curvature are widely investigated due
50 to their lightweight nature, and ability to absorb compressive energy, exchange heat, and dampen
51 acoustic vibrations [13-15]. However, the sophisticated geometry of these lattice structures has
52 proven challenging to fabricate by conventional methods. The emerging capability of material
53 extrusion additive manufacturing (MEAM) could solve some of the bottlenecks, making such lattice
54 structures attractive for different applications [16-21]. According to specific needs, with MEAM
55 these structures can be fabricated in various materials, from polymers to hydrogels, up to metal
56 alloys [22, 23]. Among this wide range of materials, those that allow the creation of soft lattice
57 structures, i.e., lattices that undergo large elastic deformations [24], can be of high interest in
58 different areas of healthcare. Their flexibility and tunability (from micro to macro-scale), could
59 make soft lattices valid substitutes for polyurethane-based foams [25], with the aim to provide
60 wearable and personalized support structures for patients with specific clinical needs. For instance,
61 as recently reported by Holmes and colleagues [26], 3D printed soft gyroid lattice metamaterials
62 can have a significant role in the treatment of decubitus ulcers or pressure injuries, as they have the
63 ability to reproduce the mechanical behavior of soft padding foams already on the market (e.g. EN
64 36-90, EN 40-230, and MA 36-600 manufactured by Dunlop Foams). In addition, the ability to
65 finely control the unit cell sizes, the surface-to-volume fraction, and the printing processes are
66 essential requirements to guarantee the fabrication of soft porous structures for 3D cell culture [27]

67 and therefore broadening the applications of soft lattices via MEAM. Such parameters ensure
68 proper cellular adhesion, colonization within the scaffolds, a good permeability of fluid media and
69 oxygen, and the possibility of vascularization, thus overcoming the current limitations of the
70 bioprinting technique [28]. Also, the feasibility of using size-programmable 3D printed soft lattices
71 could be helpful for pharmaceutical applications to test and achieve customized and controlled
72 drug-delivery systems, adapting the drug posology to the patient simply by changing the 3D design
73 of the structures [29]. Finally, the possibility of prototyping through MEAM customized and
74 tailored soft orthoses and cushions [30] could pave the way for their use in neurodegenerative and
75 neuromuscular pathologies, such as spinal cord injury (SCI), traumatic brain injury (TBI),
76 amyotrophic lateral sclerosis (ALS), and spinal muscular atrophy (SMA).

77 Nevertheless, although different design approaches have been proposed for such structures [31-36],
78 a critical challenge that still needs to be addressed for the design of lattices is to choose the
79 appropriate lattice design variables, such as cell type, unit cell size, and volume fraction in relation
80 to the selected material. Furthermore, using MEAM, there are still not enough explored
81 manufacturing constraints regarding the minimum and maximum length of the periodic minimal
82 surfaces, self-supporting sloping angles, and minimum diameters or thicknesses that lead to
83 practical limits on the shape and size of structures that can be produced using flexible materials.

84 Among these, it is important to emphasize the feasibility of producing through FFF supportless
85 lattice structures, particularly when printing hyperelastic material. As claimed in [37, 38], one of the
86 major challenges in the AM process is the removal of unwanted support structures from the lattice,
87 since they consume extra material and increase printing time, and energy for manufacturing. To
88 overcome this issue, the authors reported the design of a shell-shaped lattice structure inspired by
89 sea urchin morphology that can be additively manufactured by MEAM without requiring any
90 support structures useful for possible application in customized shoe midsoles, ski boots, tires,
91 automotive crush boxes, or any other energy-absorbing structures. They highlighted that such
92 supportless lattice did not show sagging or failure in both the experimental test and predictive

93 model, thus reducing the manufacturing and post-processing time, saving a significant amount of
94 material without compromising quality.

95 Lastly, the computational models applied to investigate the effect of cell type, the surface-to-
96 volume fraction, and the combination of different periodic surfaces (i.e., graded or hybrid) on the
97 mechanical behavior of these soft structures are still debated and poorly understood.

98 For these reasons, in this study, we investigated the capability of MEAM to produce tunable soft
99 lattice structures by combining the Functional Lattice Package (FLatt Pack) program with finite
100 element method (FEM) (1) to address design issues, (2) assess the optimum processing parameters
101 for Thermoplastic Polyurethane (TPU, 80 Shore A) soft lattices, and (3) predict their mechanical
102 behavior, useful for materials science, tissue engineering, and biomedical implants. Paying attention
103 to the available polymers for this purpose, we selected the TPU because of its high flexibility and
104 deformability. Combined with the proper lattice architecture, these characteristics might increase
105 compressive loading efficiency, energy absorption, and crashworthiness. These capabilities can be
106 relevant for developing innovative lightweight wearable solutions or scaffolds for the healthcare
107 sector. The FLatt Pack program, created by Maskery [39], possesses several peculiar features for
108 the development of lattice structures, including: (1) twenty-three lattice cell types covering a broad
109 range of pore connectivity, structural anisotropy, and surface area; (2) a graphical user interface
110 (GUI) presenting the lattice design stages in a sequential manner; (3) surface-to-volume estimation;
111 (4) relative modulus estimation; (5) the possibility of creating graded structures; (6) the option to
112 export designs in appropriate formats for 3D printing and finite element simulation.

113 Through this software tool, we generated three lattice structures that represent the focus of our
114 study; the so-called schwartz-P, gyroid, and honeycomb structures. Even if the honeycomb is not
115 classifiable as a TPMS, it was studied for the following reasons: it has a geometrically simple but
116 effective periodic shape; it is a wall-based structure which means that surfaces and not beams
117 represent its main constitutive features as in the TPMS; its stiffness and wall-based shape can be
118 attractive for the design of scaffolds for the repair of human bone defects [40].

119 Each TPU-based structure was 3D printed with different unit cell sizes, spanning from 3 to 12 mm
120 (equal to a volume fraction ranging from 0.14 to 0.56). The intent was to explore their behavior at
121 different dimensional levels and thus for multiple applications, keeping the same thickness of 1 mm
122 for all samples. To assess the printability limitation and the soft lattices manufacturability, we
123 studied the effects of the following variables: printing temperature, retraction speed, retraction
124 distance, printing speed, wipe distance, extrusion multiplier, and fan speed. Further, by using finite
125 element (FE) simulation either in linear or non-linear hyperelastic models, we examined the
126 behavior of these lattices under compressive loading. A systematic investigation into the
127 mechanical behavior of each manufactured soft lattice structure was performed experimentally to
128 validate the FE simulation data.

129 Since graded and hybrid structures are strategic for designing material-efficient solutions [41-45],
130 the optimized print parameters were used to probe the printability and mechanical behavior of a
131 gyroid with a variable volume fraction distribution and of a honeycomb with a hybrid design.

132 Ultimately, such integrated design approach, which includes FLatt Pack, FEM method, mechanical
133 testing, and 3D printing allowed us to engineer well-defined and tailorable soft lattice structures,
134 which can be used as a starting point for building innovative material-driven properties (e.g.,
135 acoustic, thermal, energetic), scaffolds for targeted tissues or 3D cell cultures, light-weighting soft
136 orthosis (e.g., non-invasive ventilation masks for mechanical ventilation, foot-beads, wrist brace),
137 and medical implants (e.g., cardiac stents, padding cushion, drug-delivery systems).

138 These thorough investigations can be a valuable strategy to effectively push the boundaries of the
139 fused filament fabrication (FFF) process for soft lattice structures. Indeed, optimizing the FFF
140 printing parameters can allow a printing resolution adequate to print flexible lattice microstructures
141 with the desired level of accuracy. Besides, it can push forwards the use of biocompatible and
142 bioabsorbable polymers, more suitable for biomedical applications, as the feedstock is safer and
143 easier to handle and requires no further post-processing. The FFF process has the chance to

144 guarantee more versatility compared to, for example, the laser sintering (SLS) or the
145 stereolithography (SLA) processes.

146

147 **2. Materials and Methods**

148 **2.1. Materials**

149 Thermoplastic polyurethane filament (TPU) 80 Shore A with a 2.85 mm diameter (FlexMark8,
150 Treed Filaments, Italy) was used as the feedstock material, without further modifications. All
151 materials were handled with gloved hands, and standard surface analysis laboratory practices were
152 followed to minimize any possible contamination.

153

154 **2.2. Design of lattice structures**

155 The honeycombs were generated using Autodesk Inventor software (Autodesk 2020, McInnis
156 Parkway San Rafael, CA, USA), selecting a wall thickness of 1 mm for each designed cell and
157 sketching a regular hexagon as base for the extrusion. Four different cell sizes were chosen for the
158 analysis, ranging from 3 mm to 6 mm. The model height has been set equal to the cell size, thus
159 increasing the model height as the cell size increases. The schwartz-P and gyroid structures were
160 generated using the Functional Lattice Package (FLatt Pack) program (2021, University of
161 Nottingham) capable of designing lattice structures starting from the size of one cell and its volume
162 fraction [39]. Before introducing the surface equations used to generate such lattice structures, we
163 introduce some terms related to their design. The first of them is the periodicity κ calculated as
164 follows:

165

$$166 \quad \kappa_i = 2\pi n_i \quad (1)$$

167

168 where i refers to the x , y , and z directions and n_i are the number of cell repetitions in each of those
169 directions.

170 A shorthand notation for sine and cosine periodic function is then defined, as follows:

171

$$172 \quad S_i = \sin \left(\kappa_i \frac{i}{L_i} \right) \quad (2)$$

173 and

$$174 \quad C_i = \cos \left(\kappa_i \frac{i}{L_i} \right) \quad (3)$$

175

176 where L_i is the absolute size of the lattice structure in the direction i of interest.

177 Then, the approximated functions to obtain the schwartz-P (U_P) and gyroid (U_G) structures are
178 calculated as follows:

179

$$180 \quad U_P = C_x + C_y + C_z - t \quad (4)$$

$$181 \quad U_G = C_x S_y + C_y S_z + C_z S_x - t \quad (5)$$

182

183 where $C_{x,y,z}$ and $S_{x,y,z}$ refer to the periodic sine and cosine function described above and t is an
184 arbitrary constant related to the volume fraction of the generated lattice structure.

185 Four different unit cell sizes were adopted (6, 8, 10, and 12 mm) and fine-tuned to obtain walls of 1
186 mm thickness. The selection of the geometry size was closely related to the ability of the FFF
187 printer to print them with the proper tuning of the selected printing parameters (see Section 2.4).

188

189 ***2.3. Design of graded and hybrid lattice structures***

190 The three-layered hybrid structure was created using Autodesk Inventor, combining honeycomb
191 oriented orthogonally with respect to the y - z and to x - y planes, in such a way that the orientation
192 was x -direction in the top and bottom thirds, and z -direction in the middle third. The unit cell size
193 was set as 6 mm, and 3×2 cells were used for the top and bottom thirds, while 5×3 cells for the
194 middle third. The thickness of the wall model was set to 1 mm and exported as “stp” file for FE

195 simulation. Instead, the graded gyroid lattice structure was generated using FLatt Pack selecting a
196 linear gradient variation of the volume fraction from 0.23 to 0.46 (corresponding to a thickness
197 variation from 1 mm to 2 mm). A $2 \times 2 \times 2$ lattice was generated and the model was exported as an
198 input file (.inp) for the FE simulations. In both cases, considering the aim of the study, we
199 controlled the model generation by monitoring the resulting thickness of the 3D model main
200 features.

201

202 ***2.4. Selection of the printing parameters***

203 All lattice structures were 3D printed using the FFF Delta WASP 2040 Industrial X (World's
204 Advanced Saving Project, Massa Lombarda, Italy) machine adopting the WASP FLEX direct drive
205 extruder suitable for flexible materials up to Shore 80 A. The printer has a maximum building
206 volume of $200 \times 200 \times 400$ mm, and a layer resolution of 50 μm , corresponding to a nozzle with 0.4
207 mm of diameter. The following parameters were kept constant: nozzle diameter (0.4 mm), infill
208 (100%), and layer height (0.15 mm). The following parameters were tuned to probe the optimum
209 printing combination: printing temperature, retraction speed, retraction distance, printing speed,
210 wipe distance, extrusion multiplier, and fan speed. Each scaffold was modeled using FLatt Pack
211 (except for honeycomb, which was modeled with Autodesk Inventor[®] software), then exported in
212 stereolithography (.stl) format and sliced using the Simplify3D[®] software. To ensure the adhesion of
213 the scaffolds to the surface of the printing platform, a 3D glue stick (Magigoo[™], Swieqi, Malta)
214 was used.

215

216 ***2.5. Finite element analysis***

217 Finite Element (FE) simulations were performed to study the mechanical behavior of the three cell-
218 based structures. The honeycomb structures were exported as “stp” files from Autodesk Inventor
219 software and then imported as continuous solid parts. Instead, schwartz-P and gyroid structures
220 obtained through the FLatt Pack program were uploaded for the FE simulations as input (.inp) files

221 (already meshed and ready to be processed, Figure S1). Two different types of simulations were
222 conducted considering the behavior of the TPU material as linear or hyperelastic. In the linear
223 elastic condition, the Young's modulus and Poisson's ratio of TPU were assumed to be 26 MPa and
224 0.45, respectively, as reported previously [46, 47]. A three-variable Mooney-Rivlin model was used
225 for nonlinear hyperelastic simulations, with a $C_{01} = 0.363$ MPa, $C_{10} = 2.93$ MPa, and $D = 0$ to
226 simulate incompressibility of the material, as previously reported [48, 49].

227 The compression simulation was performed by imposing the correct boundary conditions on two
228 planes of the imported lattice structure, particularly on two parallel faces, oriented orthogonally
229 with respect to the direction of interest to calculate the Young's modulus. A reference point was
230 applied to the fixed surface, and linked with the whole area with a tie rod type Multi-Point
231 Constraints (MPC) to correctly extract the reaction force value without further point integrations. A
232 second reference point was then linked with a tie-type MPC with the moving face (i.e., the one
233 subjected to the imposed displacement) to extract the corresponding displacement value of the
234 whole surface. The calculation of the Young's modulus was performed as follows:

235

$$236 \quad E_i = \frac{\sigma_i}{\varepsilon_i} \quad (6)$$

237

238 where E is the equivalent Young's modulus in the direction i of interest; σ is the equivalent stress,
239 calculated as the ratio between the reaction force and the equivalent area; ε is the strain, calculated
240 as the ratio between the imposed displacement and initial length of the lattice in the direction i .

241

242 The boundary conditions were applied through the two reference points. An encaster is applied to
243 the first reference point. For the second reference point an initial displacement of 5% of
244 compressive strain in the direction of interest was applied.

245 The hybrid honeycomb structure was imported as a single continuous part. For a faster calculation,
246 the model was cut in a quarter of the whole piece, to take advantage of the geometrical symmetry;

247 two shell parts were also generated to simulate a compression machine moving and fixed surfaces.
248 The two planes were treated as 2D discrete rigid parts; thus no material selection was necessary.
249 Then, boundary conditions were applied to the whole model, imposing a vertical displacement of
250 the upper part equal to 40% of the model height and an encaster constraint in the bottom part. The
251 symmetries of the simplified model need to be represented with two additional boundary
252 conditions, in the x and z directions, to allow the model to behave like the not-simplified structure.
253 See Supporting Information for further details.

254

255 ***2.6. Mechanical testing***

256 Quasi-static compression tests were performed using the MTS Synergy 200 testing machine
257 equipped with a 1 kN load cell to experimentally characterize the scaffolds' behavior and validate
258 the numerical models. Three repetitions were tested for all homogenous samples, while one sample
259 was analyzed for the hybrid honeycomb and the graded gyroid. All specimens were tested with a
260 crosshead speed of 5 mm/min.

261

262 ***2.7. Scanning electron microscopy (SEM)***

263 **Sagging, improper layer adhesion, and porosity of 3D printed honeycomb, schwartz-P, and gyroid**
264 **structures were observed using a field emission scanning electron microscopy (Zeiss EVO 50)**
265 **operating at 30 kV. All specimens were sputter coated with gold prior to examination to ensure**
266 **better conductivity and prevent the formation of electrostatic charges.**

267

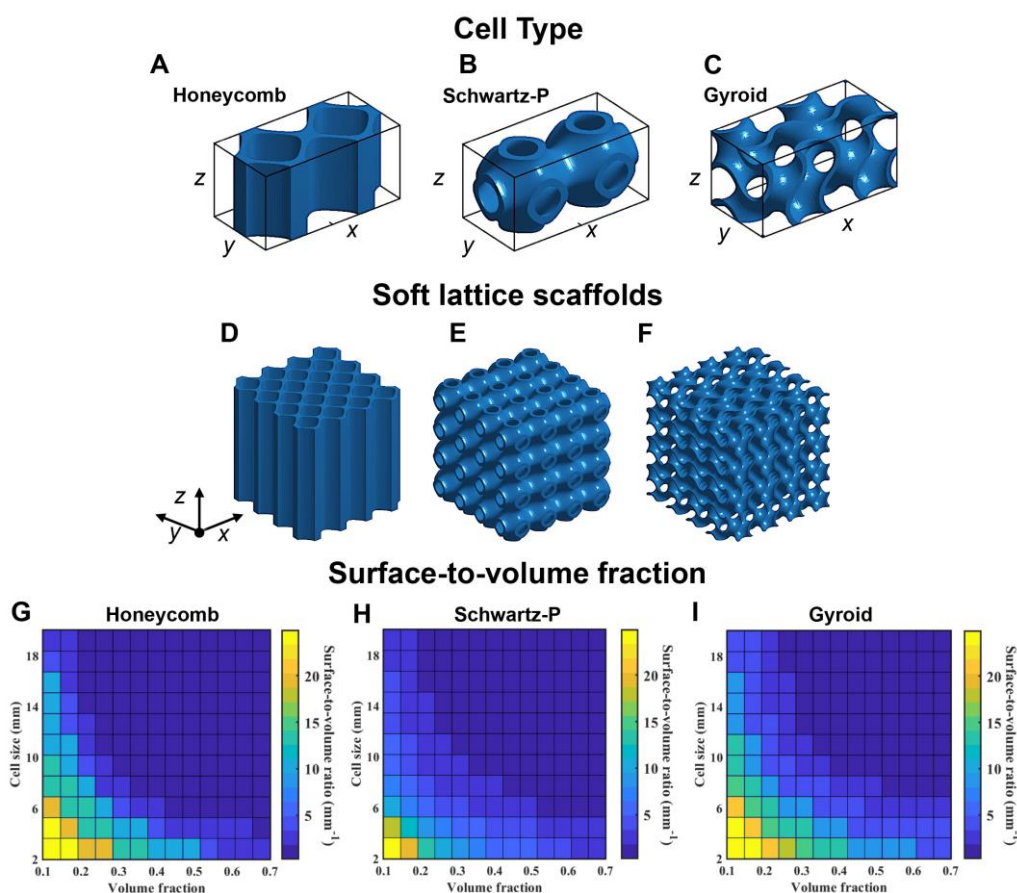
268 **3. Results and Discussion**

269 ***3.1. Soft lattice structures design and printability***

270 The honeycomb, schwartz-P, and gyroid unit cells were generated as periodic matrices [50] (Figure
271 1 A-C). Then, they were converted to three-dimensional cuboid structures starting from the size of
272 one cell and its volume fraction (Figure 1 D-F) (see Section 2.2 for further details). Furthermore,

273 taking advantage of the FLatt Pack "surface-to-volume estimation" feature, it was possible to
 274 estimate the surface-to-volume ratio for each cell over a range of volume fractions and cell sizes
 275 (Figure 1 G-I). This feature can provide a relevant design criterion not only to fine-tune the
 276 structures, but also when the intended application for the selected lattice involves the interaction of
 277 its surface with a specific environment (e.g., for biological scaffolds, where the amount of surface
 278 determines the number of cells that can attach and grow there, or for medical implants where heat
 279 transfer across the surface to a fluid medium must be ensured).

280



281

282

283 **Figure 1.** Unit cell geometry based on (A) honeycomb, (B) schwartz-P, (C) gyroid, and (D-F) corresponding cuboid
 284 matrix. (G-H) The surface-to-volume ratio for each tested cell obtained through FLatt Pack "surface-to-volume
 285 estimation" feature.

286

287 According to the literature due to topological constraints [51], the minimum unit cell size allowed

288 for printing honeycomb is 3 mm (volume fraction of 0.56), whereas for schwartz-P and gyroid is 6

289 mm (volume fraction of 0.28 and 0.46, respectively). Instead, the maximum value of unit cell
 290 achievable is 6 mm for honeycomb, and 12 mm for schwartz-P and gyroid. Then, once the
 291 preliminary unit cells sizing has been tuned, such lattices were 3D printed. An overview of the cell
 292 sizes and related volume fractions is provided in Table 1.

293

Honeycomb			Schwartz-P			Gyroid		
Unit cell size (mm)	Volume fraction	Void fraction	Unit cell size (mm)	Volume fraction	Void fraction	Unit cell size (mm)	Volume fraction	Void fraction
3	0.56	44%	6	0.28	72%	6	0.46	54%
4	0.44	56%	8	0.21	79%	8	0.35	65%
5	0.36	64%	10	0.17	83%	10	0.28	72%
6	0.31	69%	12	0.14	86%	12	0.23	77%

294 **Table 1.** The unit cell sizes allowed for the 3D printing of honeycomb, schwartz-P, and gyroid lattices, and the
 295 corresponding volume fraction and void fraction percentage values.

296

297 Based on many tests carried out, all structures were successfully 3D printed with an optimized
 298 nozzle temperature of 230 °C (Figure S2). The optimized parameters for the honeycomb with a unit
 299 cell size of 3 mm were an extrusion width of 0.65 mm, a retraction distance of 1 mm, and a
 300 retraction speed of 2,400 mm/min. Details concerning the printing path of honeycomb with a unit
 301 cell size of 3 mm are provided in Figure S3. A greater retraction distance was not achievable due to
 302 the direct drive extruder used to print the TPU through the Delta WASP 2040 Industrial X (see
 303 Section 2.4); instead, a lower value resulted in debris and lumps of filaments inside the honeycomb
 304 cells, thus turning in a reduced surface quality of the structure. The optimum printing speed was
 305 1,100 mm/min, as higher speeds did not allow proper material deposition and distorted honeycombs
 306 were generated. However, we must recall that using a slower print speed while ensuring good
 307 printability increases production times considerably. Lastly, to promote good adhesion of the first
 308 layer, a build plate temperature of 50 °C was selected, and the cooling system was turned “off”, as
 309 we observed that the contribution of the fan during the printing process caused early cooling of the
 310 TPU filament, leading to the layer’s detachment.

311 For unit cells from 4 mm up to over 6 mm, we used the same process parameters as the 3 mm unit
 312 cell size. In these cases it was possible to increase the printing speed up to 3,000 mm/min because
 313 (as shown in Figure 1 G, and Table 1) increasing the unit cell size decreases the volume fraction. It

314 is thus possible to obtain a good surface quality of the structure without debris and lumps of
315 filaments (Figure S2A), and, at the same time, decrease production times.

316 For schwartz-P the smaller unit cell successfully 3D printed was 6 mm (Figure S2B), with a
317 retraction speed of 3,000 mm/min to avoid debris on the structure's surface and a printing speed of
318 1,000 mm/min. We noticed that a faster printing speed resulted in an incorrect material deposition
319 during the bridging phase, thus generating voids and weakening the overall structures. Conversely
320 to the honeycomb structures, to promote a good adhesion of the schwartz-P structures to the build
321 plate, the first layer speed was decreased to 50% and a 120% layer width was selected; also, a 20%
322 fan speed was used because it avoided the local structure overheating. As schwartz-P geometry
323 requires the formation of small bridges between cells, the wipe distance of 2 mm length was
324 enabled to prevent additional debris during printing. Using this method, the nozzle could travel for
325 an extra 2 mm length on the same path, avoiding the spread of the material over the entire lattice's
326 surface.

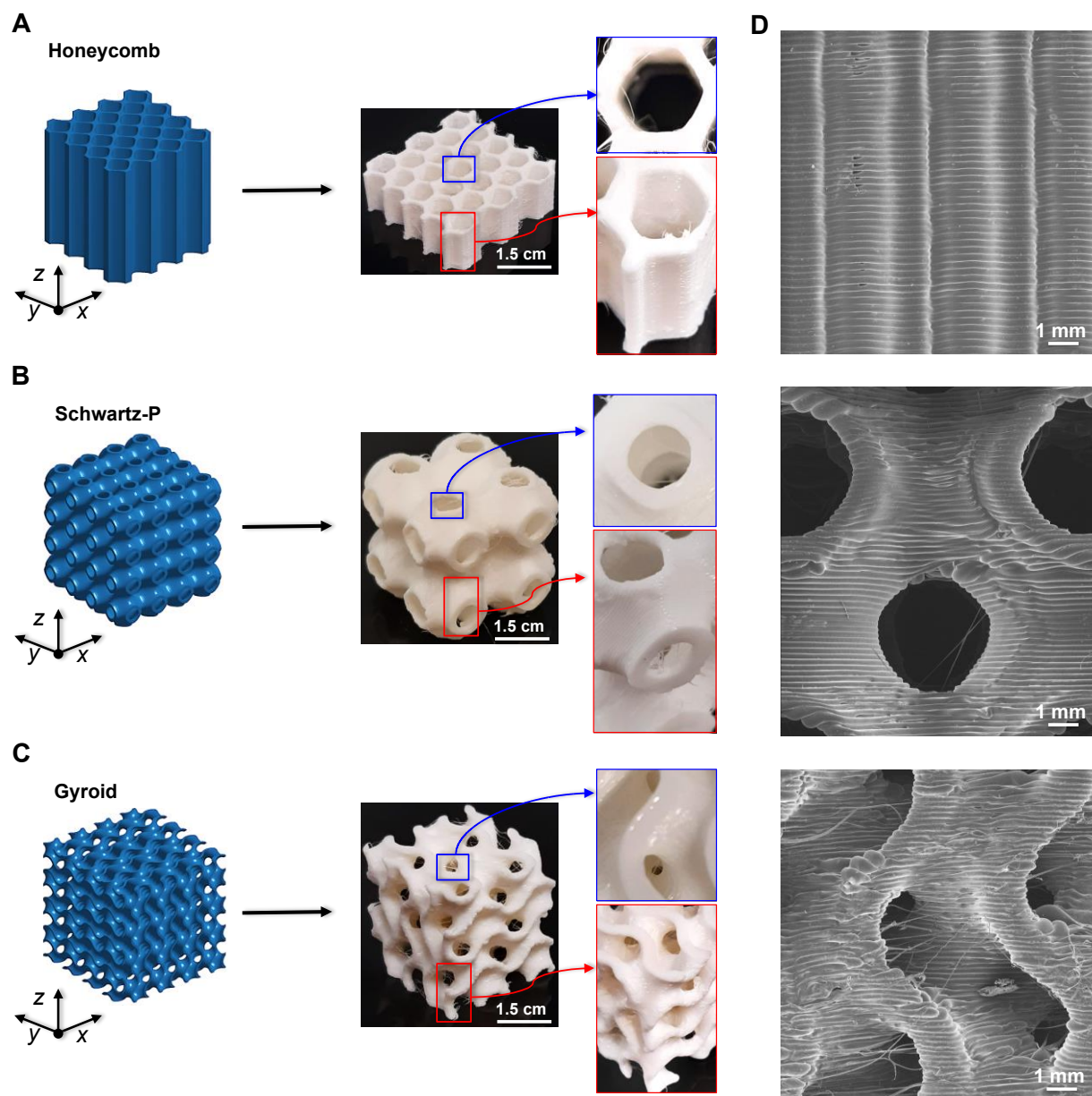
327 Also in this case, as for the honeycomb structures, with the increase in the unit cell size, it was
328 possible to increase the printing speed up to 2,400 mm/min to obtain the same result in terms of
329 surface quality of lattices. However, some drawbacks occurred during the bridging procedure (not
330 detected for 6 mm cells) for much larger unit cells due to the loss of material during the creation of
331 the cantilever structure. To overcome this issue, the extrusion multiplier was set to 1.1 while
332 keeping the print speed unchanged, thus allowing building support-less schwartz-P structures.

333 The same printing parameters used for schwartz-P were used for the gyroid structures with the unit
334 cell of 6 mm, except for the wipe nozzle option in which a wipe distance of 0.5 mm was used to
335 avoid loss of filaments on all surfaces of the final construct. Once again, for cells larger than 6 mm,
336 the print speed was increased up to 2,400 mm/min, with no surface or geometric quality loss
337 (Figure S2C). However, unlike the schwartz-P structures, there was no need to use the extrusion
338 multiplier when creating the cantilever walls of the cells. For the sake of clarity, an overview of the
339 used printing parameters is provided in Table S1.

340 Based on an initial visual inspection and as shown in Figure 2 A-C, the tuned printing parameters
341 led to satisfactory results concerning the overall printing quality of the samples considering the
342 flexible nature of the filament and the complexity of the structures. Looking at the upper surface of
343 all lattice structures, a uniform way of filling can be seen without debris, stringing inside the unit
344 cell, and long lumps of filaments. In addition, there are no visible voids in the sidewalls or other
345 imperfections and irregularities of the layers after the manufacturing process. As a reference, the
346 weights of the soft lattice structures were recorded after 3D printing and compared with the
347 theoretical values. These data were used to quantitatively analyze any differences between the
348 printed samples and corroborate the qualitative evaluations of the printing quality (Figure S4). The
349 numerical results approximately coincided with the experimental data, and the coefficient of
350 variation (COV) of this latter was 0.87%, 3.12%, and 2.54% for honeycomb, schwartz-P, and
351 gyroid structures, respectively. To seek accuracy, since the experimental weights of the gyroid
352 lattices were slightly lower than the theoretical values, the structures were analyzed through optical
353 microscopy. The external surfaces did not show defects or irregularities, so the lattices were cut in
354 half with a scalpel (to not alter the 3D printed layers) to evaluate the internal parts: some
355 “micrometric cavitations” emerged with an average value of 380 μm (Figures S5). These voids
356 could lead to a decrease in the structure’s overall weight but can also result in a reduction of
357 mechanical behavior (no micrometric cavitations were identified in the honeycomb and schwartz-P
358 structures, Figure S5). **These data were also confirmed by SEM analysis (Figures 2D). The 3D
359 printed honeycomb and schwartz-P structures did not exhibit defects such as sagging, improper
360 layer adhesion or porosity (Figure S6A-B). On the contrary, microscopic pores were instead present
361 in the gyroid structures (Figure S6C), which might adversely affect the mechanical performance of
362 the TPU-based structures.** Unfortunately, this factor goes beyond the accuracy of the design and the
363 fine-tuning of the printing parameters, but must be considered for specific applications.

364 Nevertheless, good reproducibility of the specimens was obtained. Such results emphasize the
365 efficiency and potential scalability of the FFF process to produce controlled soft lattice structures
366 with different unit cell sizes, lattice types, scales, and associated parameters.

367



368

369 **Figure 2.** FLatt Pack design matrix lattices and corresponding 3D-printed (A) honeycomb, (B) schwartz-P, and (C)
370 gyroid matrix soft scaffolds built with the TPU filament. The magnification of the upper surface (in blue) and sidewalls
371 (in red) shows the printing quality of all soft lattices without any visible voids, imperfections, or irregularities after the
372 manufacturing process. (D) SEM images of the TPU-based lattice structures. No visible defects, sagging, and improper
373 layer adhesion are observed in all honeycomb and schwartz-P specimens. Microscopic pores are present in the gyroid
374 structures.

375

376 **3.2. FEA simulations of compressive behavior for the TPMS and honeycomb soft structures**

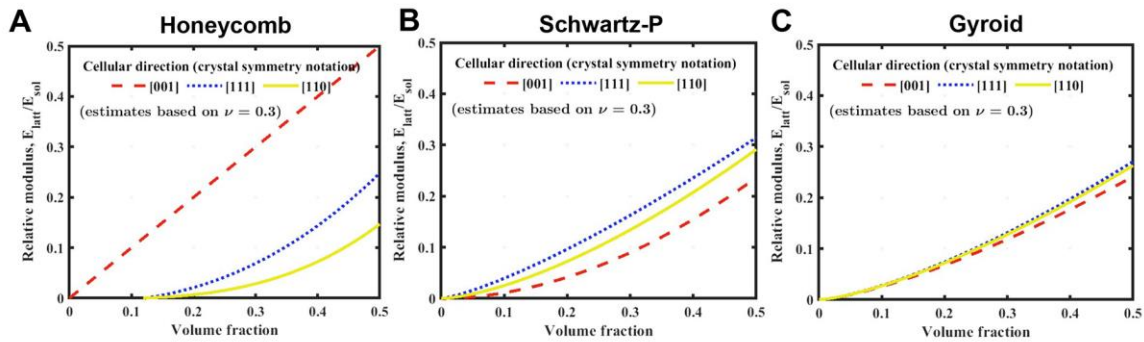
377 To gain insights into the mechanical behavior of each designed structure, we used both FLatt Pack –
378 Relative Modulus Estimation and FE models. Using FLatt Pack it was possible to explore the
379 relative elastic modulus against volume fraction for selected cell types along the x , y , and z
380 directions [52, 53]. This information allows one to select the cell type, volume fraction, and
381 orientation most suitable for the intended application, and have a preliminary and dimensionless
382 estimate of the behavior of the structure before the use of FE analysis. We noticed how stiffness
383 increases for all tested structures by increasing the volume fraction (Figure 3 A-C). In contrast, the
384 relative elastic modulus tends to reach a plateau below the 0.2 volume fraction (equal to a unit cell
385 size of 6 mm), confirmed by other previous studies [54, 55]. However, while for the schwartz-P and
386 gyroid structures the relative modulus appeared similar in the x , y , and z directions with an
387 exponential increment, it was not so for the honeycomb structures. The latter displayed an
388 exponential increase in the x and y directions with small variations of relative modulus values,
389 while in the z direction, it was possible to observe a linear increase with high values of the relative
390 modulus. This increase is due to the orientation of the vertical walls of the honeycomb structure,
391 which contributes to the model's overall rigidity and makes the honeycomb topology, along this
392 direction, a stretching-dominated structure [56].

393 Next, we examined this behavior on all structures using linear FE analysis. Using the FE
394 simulation, the amount of applied force was obtained, and the equivalent compressive elastic
395 modulus was then calculated as the ratio of compressive stress to compressive strain in the x , y , and
396 z direction of interest (see Equation (6)). The obtained elastic moduli showed the variability of the
397 mechanical properties of each lattice with the unit cell sizes along the x , y , and z directions (Figure
398 3 D-F). These results agree with the dimensionless data obtained through the FLatt Pack software.
399 The different structures were expected to have a wide range of direction-dependent compressive
400 elastic moduli (Table 2), controlled by the design parameters such as cell type, the surface-to-
401 volume fraction, and the length of the periodic minimal surfaces. The distribution of von Mises
402 stress for comparing honeycomb, schwartz-P, and gyroid structures is shown in Figure 4.

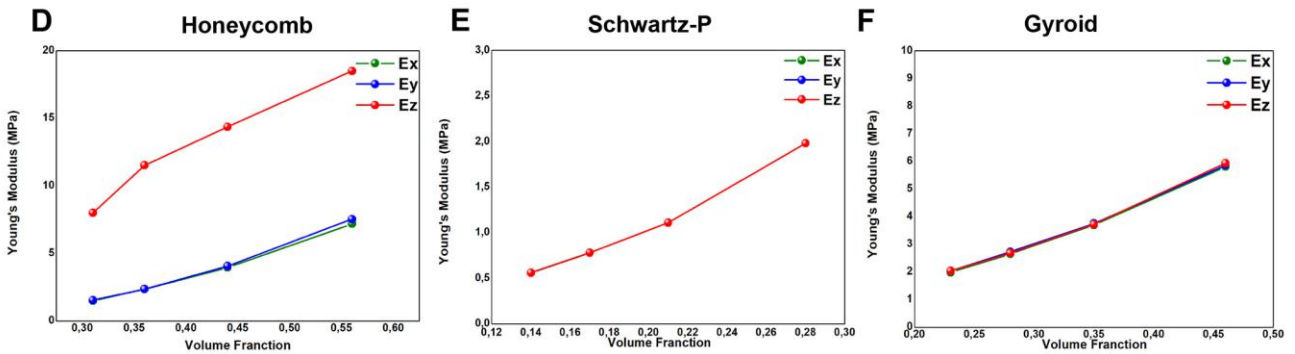
Honeycomb				Schwartz-P				Gyroid			
Unit cell size (mm)	E_x (MPa)	E_y (MPa)	E_z (MPa)	Unit cell size (mm)	E_x (MPa)	E_y (MPa)	E_z (MPa)	Unit cell size (mm)	E_x (MPa)	E_y (MPa)	E_z (MPa)
3	7.53	7.18	18.49	6	1.98	1.98	1.98	6	5.86	5.80	5.93
4	4.06	3.96	14.36	8	1.11	1.11	1.11	8	3.75	3.70	3.72
5	2.36	2.35	11.51	10	0.78	0.78	0.78	10	2.72	2.65	2.68
6	1.53	1.49	7.99	12	0.56	0.56	0.56	12	2.03	1.98	2.03

404 **Table 2.** Comparison of compressive elastic moduli in the x , y , and z directions obtained from the FE linear model of
 405 honeycomb, schwartz-P, and gyroid matrix soft lattice scaffolds with different unit cell sizes.
 406

Flatt Pack – Relative Modulus Estimation



FEM Linear – Modulus Estimation



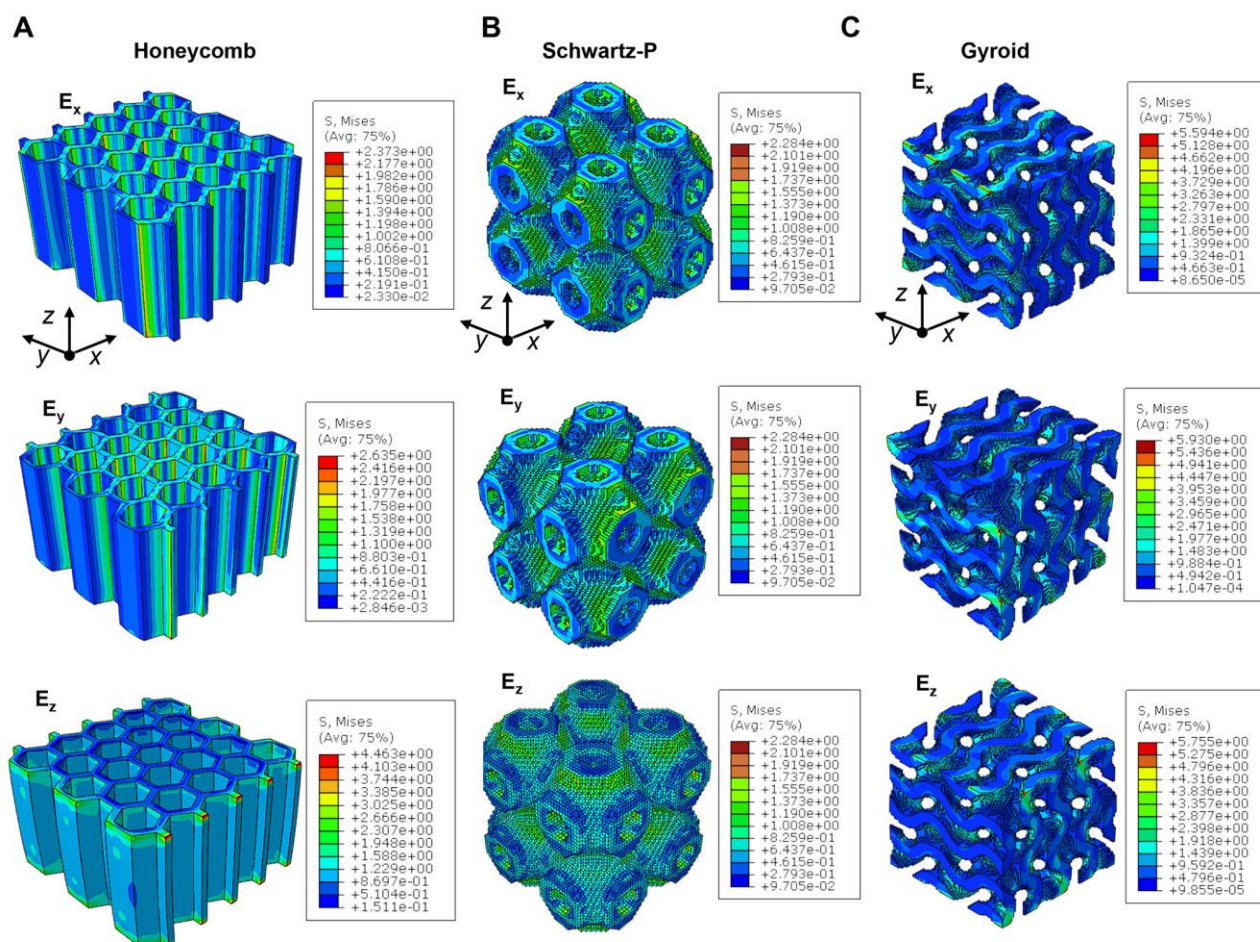
407

408 **Figure 3.** Comparison of (A-C) FLatt Pack – Relative Modulus Estimation and (D-F) linear FE models to explore the
 409 elastic modulus against volume fraction for selected cell type along the x , y , and z directions. In (E) all curves perfectly
 410 overlap.
 411

412 In detail, the analysis of the honeycomb structures confirmed that Young's moduli in the x and y
 413 directions have the same value, with minor variations probably related to the cells repetition,
 414 characterized by half the original thickness at the upper and lower parts of the scaffold when
 415 creating the 3D model. Instead, as expected, the Young's moduli in the z direction (i.e., out-of-plane
 416 direction) showed values 10-fold higher than those for the in-plane moduli. The out-of-plane
 417 direction results were always stiffer than in-plane directions due to the orientation of the vertical

418 walls, which contribute to the overall stiffness of the model (Figure 4 A). Conversely, we observed
 419 that the values of Young's moduli for schwartz-P and gyroid structures have no directional
 420 dependence, probably due to the symmetry in their geometries (Figure 4 B-C). Also, the increase in
 421 the unit cell size, maintaining the same thickness of the model, led to a decrease in the volume
 422 fraction and consequently a reduction in the value of the compressive modulus.

423



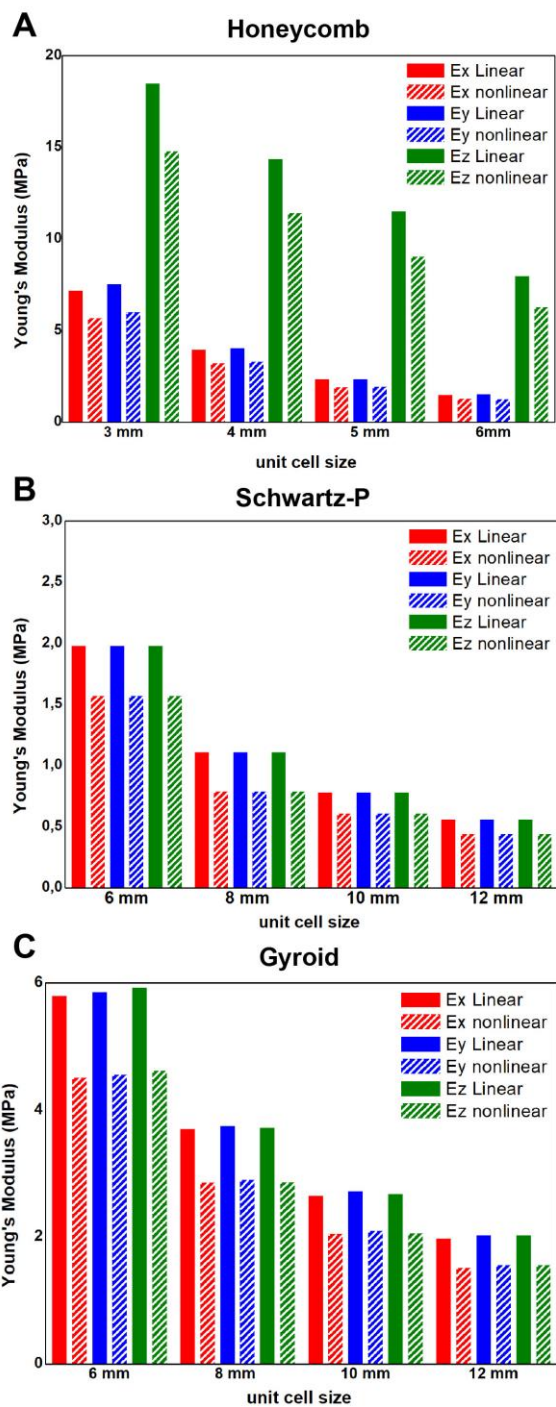
424

425 **Figure 4.** Von Mises stress distribution of soft lattice scaffolds with different geometry derived from FE models under
 426 compression along the x , y , and z directions.

427

428 Lastly, we sought to estimate the compressive behavior of such soft lattices via nonlinear FEM
 429 analysis (Figure 5), using a three-variable Mooney-Rivlin hyperelastic model (see Section 2.5 for
 430 further details), as this model would better describe the material behavior of TPU filament. The
 431 results showed an overall decrease in the compressive modulus of each tested structure compared to

432 linear results; in particular a reduction of 23% is registered for schwartz-P and gyroid structure,
 433 among all unit cell sizes. This was not applicable for honeycomb structures, where differences in
 434 compressive modulus tend to be smaller as the unit cell size increases (it goes from 20% for the unit
 435 cell size of 3 mm, to 13% for the unit cell size of 6 mm). Additionally, based on the hyperelastic
 436 model, the honeycomb structures were stiffer than the schwartz-P and gyroid structure (Table S2).
 437



439 **Figure 5.** Comparison of linear (striped columns) and hyperelastic (full columns) FEM-derived compressive moduli
440 along the x , y , and z directions of (A) honeycomb, (B) schwartz-P, and (C) gyroid matrix soft lattice scaffolds with
441 different unit cell sizes.
442

443 These results may provide the basis for implementing a non-dimensional and dimensional data
444 coupling-based approach for boosting the understanding of soft lattices in additive manufacturing
445 design. Although theoretical, this approach can provide a deeper understanding of the structure-
446 mechanical property relationship of each cell family that can be used to design soft innovative-
447 engineered materials models with tunable properties. However, we must emphasize that the
448 comparison between FE simulations and the additively manufactured structures may differ. Due to
449 the intrinsic nature of the 3D printing process, the structure is generated via superimposition of
450 subsequent layers, while the FE model is characterized as a continuous homogeneous solid with
451 isotropic features. Hence, the influence of the layer bonding is not considered in the FE model, thus
452 leading to be a key deviation to which attention must be paid when comparing theoretical versus
453 experimental behaviors of 3D-printed structures.

454

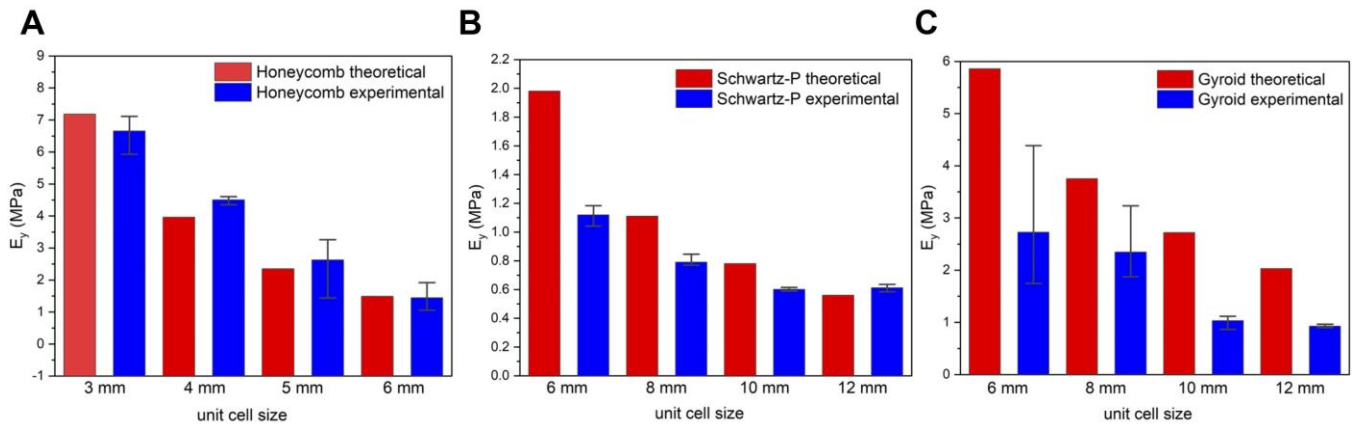
455 ***3.3. Experimental results for the homogeneous soft structures***

456 The results of the quasi-static compression tests are summarized in Figures 6 A-C, for the
457 honeycomb, schwartz-P, and gyroid structures, respectively. The moduli were calculated based on
458 Equation (6). For the sake of clarity, the theoretical modulus referred to the testing direction and
459 obtained from the numerical analyses is provided. This value is the E_y (Table 2).

460 The honeycomb samples confirm what was obtained from the numerical analyses. These samples
461 are characterized by a mass gain (Figure S4), which increases the E_y value. The only exception is
462 the structure with a unit cell size of 3 mm, which, despite the mass increase (Figure S4), has an E_y
463 value slightly lower (6.6 ± 0.52 MPa) than the theoretical one (7.18 MPa). This behavior could be
464 due to some minor irregularities on the printing walls.

465 For the schwartz-P, despite the increase in mass, which characterizes the 3D printed samples, we
466 recorded an E_y value consistently lower than the numerical values. The mass gain is due to the
467 change in the extrusion multiplier parameter set to 1.1 to guarantee the printability of the structure.
468 Only when this mass increase significantly overcomes the theoretical value (i.e., in the lattice with a
469 unit cell size of 12 mm, Figure S4), there is an opposite situation among theoretical and numerical
470 values. This difference is due to the mesh used for simulating the behavior of the schwartz-P
471 structures. As shown in Figure 4, the selected mesh tends to overestimate the dimensions of the
472 structure, especially on the sloped surfaces. For the 12 mm unit cell size structure, this
473 overestimation almost compensates the increase in the mass value, probably because there is a
474 larger volume fraction and therefore more finite elements in proportion. The mass gain is less
475 evident in the case of 6 mm unit cell because the printing path leaves material gaps at the bridges,
476 which connect the unit cells (Figure S7). This issue does not occur in the other samples schwartz-P
477 samples (Figure S7).

478 Finally, the gyroid demonstrated the highest discrepancy between theoretical and experimental
479 values for all the tested sizes (Figure 6C). An experimental modulus lower than the theoretical one
480 was expected considering the mass deficit, which characterizes these samples (Figure S4), as also
481 demonstrated by the presence of micrometric cavitations (Figure S5) and the material gaps which
482 are present at the connection zones (i.e., the bridges) of unit cells in the case of scaffolds of 6 mm
483 unit cell dimension (Figure S7). Indeed, the influence of the mass deficit on the mechanical
484 behavior of the structures is even more evident in the structure having a 6 mm unit cell size. This
485 structure has the highest volume fraction (i.e., 0.46, Table 1), and it is also bending-dominated, as
486 demonstrated by Maskery et al., [57]. Hence, the scaling-law that describes the influence of the
487 volume fraction on the relative elastic modulus (i.e., the modulus of the lattice divided by the
488 modulus of the material it is made from) has a coefficient equal to 2 [56], which means that
489 especially at high-volume fractions, the relative modulus of the structure is more sensitive to
490 volume-fraction changes.



492

493 **Figure 6.** Comparison among theoretical and experimental Young's Moduli of the analyzed structure: (A) honeycomb,
 494 (B) schwartz-P, and (C) gyroid. According to the selected testing direction, the theoretical values are the E_y provided in
 495 Table 2.

496

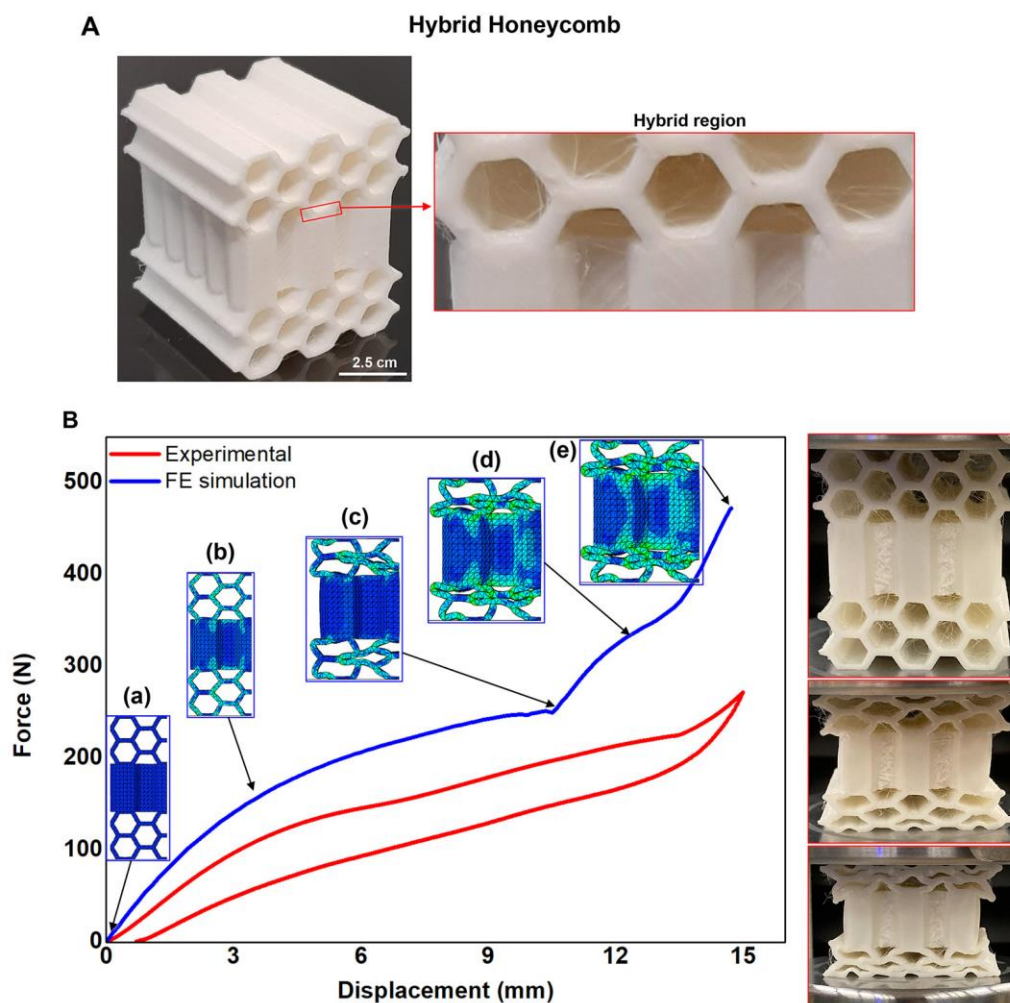
497 3.4. Graded and hybrid soft structures

498 Lastly, we explore two further possibilities of lattice structure design using the optimized print
 499 parameters: the gyroid volume fraction grading and the honeycomb hybridization. Both can provide
 500 novel soft lattice behavior, such as layer-by-layer structural collapse and tailorable impactor
 501 deceleration under dynamic loading. Once designed, the honeycomb three-layered hybrid structure
 502 (see section 2.3) was 3D printed using the optimized process parameters for a unit cell of 6 mm
 503 (Figure 7A, and Table S1). Then, we examined its compressive behavior using linear FEM analysis
 504 imposing a vertical displacement of the upper part equal to 40% of the model height and an encaster
 505 constraint in the lower part (see Section 2.5 for further details). The FEM deformation of the three-
 506 layer hybrid honeycomb lattice under compressive loading with the corresponding force-
 507 displacement curve is shown in Figure 7B. At low strain, all of the observed deformations were in
 508 the low-stiffness x direction regions, and the initial elastic response and plastic plateau are identified
 509 as (a) and (b) on the force-displacement curve. At the phase identified as (c), the cell walls were
 510 completing their bending into the stiffer configuration, reaching above (d) an extremely stiff
 511 configuration as the x regions at the bottom and top of the structure entered densification. The
 512 densification mechanism exists for all cellular structures at high strain, but in this case, it was

513 localized to the x regions only. Finally, the three-layer hybrid lattice underwent a further increase in
514 stiffness (e) as the compressive load was transferred solely through the remaining high-stiffness
515 region z direction. The experimental force-displacement curve (Figure 7B) confirms the trend
516 obtained through the numerical analysis, as demonstrated by the sample images at different test
517 phases. However, there is a general overestimation of the force values: it might be due to the
518 applied boundaries conditions, which lead to over constraining the overall structure.

519 In addition, it should be noted that the FE analysis conducted on the hybrid structure takes into
520 account some approximations in order to allow the convergence of the simulation (see section 2.5);
521 however, we do not exclude that in future studies it will be possible to apply the advanced material
522 model aimed at studying the hysteretic behavior of the hyperelastic TPU polymer bioinspired by the
523 morphology of the sea urchin for complex soft hybrid structures, as recently reported in [58].

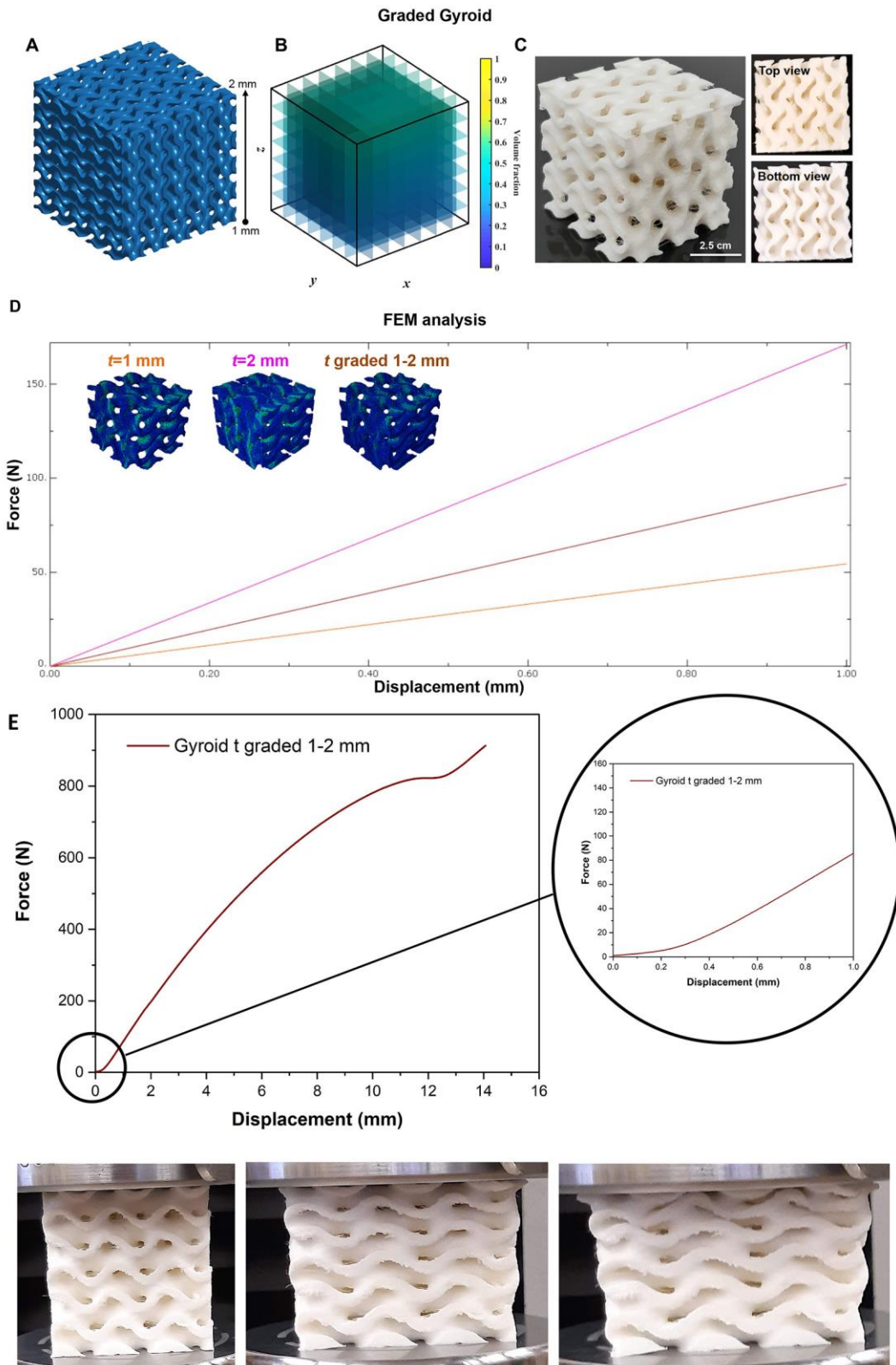
524



525

526 **Figure 7.** (A) 3D printed three-layer hybrid honeycomb soft lattice. In red magnification of the hybrid region, which
527 shows how the two regions are well joined together with no visible voids. (B) Numerical and experimental force-
528 displacement curves of the three-layer hybrid honeycomb lattice under compressive loading, which show: (a) initial
529 elastic response and (b) plastic plateau of the hybrid soft lattice; (c) bending phase of cell walls; (d) stiffer configuration
530 as the regions at the bottom and top of the structure entered densification; (e) further increase in stiffness as the
531 compressive load transfers solely through the remaining high-stiffness region (i.e., in the z-direction). On the left, a
532 sequence of photos taken during the experimental tests at different compression stages.
533

534 The soft gyroid graded lattice was generated using FLatt Pack, by selecting a linear gradient
535 variation of the volume fraction from 0.23 to 0.46 (corresponding to a thickness variation from 1
536 mm to 2 mm). A $2 \times 2 \times 2$ lattice was generated for the FE simulations (Figure 8 A-B) and 3D
537 printed using optimized parameters (Figure 8C, and Table S1). The compression simulation was
538 performed with the imposed displacement oriented as the direction selected for increasing the
539 thickness. The corresponding force-displacement curve was compared with the curves of the
540 structures having a uniform thickness of 1 mm and 2 mm (Figure 8D), respectively. FE simulations
541 displayed that in the displacement region analyzed, the soft gyroid graded lattice had an
542 intermediate behavior compared to those with a uniform thickness of 1 mm and 2 mm. However, it
543 was impossible to go beyond a 1 mm displacement due to computational limits. This would have
544 allowed evaluating whether cell walls between 1 and 1.5 mm completed their folding in the most
545 rigid configuration and entered densification, then undergoing a further increase in stiffness due to
546 the remaining 2 mm thick region. This computational bottleneck was overcome by testing the 3D-
547 printed sample. As shown in Figure 8E and specifically in the zoomed image, the value of the force
548 at 1 mm displacement (90 N) is similar to the theoretical one (95 N). As expected, the increase in
549 the structure's stiffness is instead shown on the left of Figure 8E. There is a monotonic increase in
550 the stiffness until the densification phase is reached. The bottom of Figure 8E shows the sample at
551 different compression stages.



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Figure 8. (A) Matrix representation of soft graded gyroid lattice, (B) and corresponding linear volume fraction preview. (C) Soft graded gyroid printed sample. (D) Force-displacement curve derived from FEM deformation of the soft graded gyroid lattice (in brown) under compressive loading, compared with lattices with a uniform thickness of 1 mm (in orange) and 2 mm (in magenta). (E) The experimental force-displacement curve and a magnification of the same chart from 0 to 1 mm of displacement. At the bottom is the sample during different compression stages.

560 **4. Conclusion**

561 The study describes the numerical and experimental analyses performed to explore the behavior of
562 a selection of lattices, such as the honeycomb, the schwartz-P, and the gyroid 3D-printed using
563 thermoplastic polyurethane (TPU) as feedstock material. These soft lattices can be relevant for
564 multiple purposes, especially in the biomedical field. They could be used to develop biological
565 scaffolds, biomedical implants, porous structures for 3D cell culture, soft orthosis, and soft cushions
566 to control pressure ulcers or bedsores in patients with immobility who are forced to use a
567 wheelchair. They could also be used in all circumstances where there is semi-permanent contact
568 with the skin, such as lower limb prostheses or masks for non-invasive ventilation (NIV) for severe
569 acute respiratory syndrome coronavirus (SARS-CoV) or neuromuscular patients. However, despite
570 this wide range of possible applications, there are still not enough studies exploring their behavior
571 and technical feasibility concerning printing-related issues. This paper overcomes this gap by
572 proposing a collection of design and printing guidelines for fabricating these structures using the
573 fused filament fabrication (FFF) process. The selection of this printing process was made on
574 purpose, considering its wide diffusion and flexibility in terms of materials that could be potentially
575 3D-printed, for example, combining the thermoplastic matrix with specific additives tailored for the
576 biomedical application of interest. Multiple design and printing parameters were explored. We
577 started defining the main design variables for the scaffolds, such as the cell type, the unit cell size,
578 and the volume fraction. These variables were combined with the selected printing parameters to be
579 tuned to obtain self-supporting, stable structures and repeatable printing results.

580 In particular, we optimized the multi-scale printing parameters, noting that for the honeycomb
581 structure with a unit cell size of 3 mm, to achieve a good surface quality of the structure without
582 debris and lumps of filaments, an extrusion width of 0.65 mm, a retraction distance of 1 mm, a
583 retraction speed of 2,400 mm/min, and a printing speed of 1,100 mm/min must be applied. On the
584 other hand, for unit cells from 4 mm up to over 6 mm, the same process parameters as the 3 mm
585 unit cell size could be used, but with a print speed of up to 3,000 mm/min. Instead, for the TMPS

586 schwartz-P and gyroid structures with a smaller unit cell of 6 mm, the same printing parameters can
587 be applied, namely: a retraction speed of 3,000 mm/min to avoid debris on the surface of the
588 structure and a printing speed of 1,000 mm/min; with the increase of the unit cell size it was
589 possible to increase the printing speed up to 2,400 mm/min to obtain the same result in terms of
590 surface quality of the gratings. However, unlike the gyroid structures, for the schwartz-P structures
591 it was necessary to use the extrusion multiplier when creating the cantilevered cell walls.

592 In parallel, numerical analyses were performed to understand the scaffolds' behavior with the
593 selected design parameters in linear and hyperelastic conditions, to explore which model would
594 better describe the material behavior of TPU filament. These analyses were validated
595 experimentally using quasi-static compression tests, which showed a good matching with the
596 theoretical data. Together with homogenous samples, i.e., scaffolds having a constant thickness, we
597 also successfully explored the effectiveness of the optimized printing parameters in case of a not-
598 constant thickness (i.e., graded) and hybrid structures.

599 The obtained results are promising, but further research efforts are needed to extend more and more
600 research opportunities in the design and printing of soft lattices, considering the wide range of
601 applications in which they could be employed.

602

603 **Declaration of Competing Interest**

604 The authors declare no competing interest.

605

606 **Author Contributions:** “Conceptualization, R.P.; methodology, R.S., S.G., and R.P.; software,
607 R.S., and R.P.; validation, R.S., S.G., and R.P.; formal analysis, R.S. and R.P.; investigation, R.S.,
608 S.G., and R.P.; data curation, R.S. and R.P.; writing—original draft preparation, S.G., and R.P.;
609 writing—review and editing, R.S., S.G., S.R., and R.P.; visualization, R.S., S.G., S.R., and R.P.;
610 supervision, R.P.; project administration, R.P.; funding acquisition, S.R. and R.P.. All authors have
611 read and agreed to the published version of the manuscript.”

612

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616

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796 *Supplementary information for*

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798 **Insights into the printing parameters and characterization of**
799 **thermoplastic polyurethane soft triply periodic minimal surface and**
800 **honeycomb lattices for broadening material extrusion applicability**

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810 **2.5. Finite element analysis**

811 In the STEP section of the models, the option Nlgeom was set to “on” for allowing non-linearities

812 in the models during compression simulations. The maximum number of increments was set to 100,

813 with an increment size variable from an initial 0.01 up to 0.1. The minimum increment size was left

814 to a default value of 1.0×10^{-5} . The output requested for such simulations were the stress

815 distribution, the reaction forces, and the displacements of reference points selected and highlighted.

816 The 3D eight-node linear isoparametric elements (C3D8H) were adopted for all models; no reduced

817 integration was applied, but a hybrid formulation command was required due to the

818 incompressibility of the simulated TPU. The average mesh size selected for this type of lattice was

819 0.25 mm, except for the 6 mm cell size where the average finite element size was set to 0.4 mm to

820 avoid a too long computation time.

821 For the hybrid honeycomb in the STEP section, the non-linear geometry option was turned “on”,

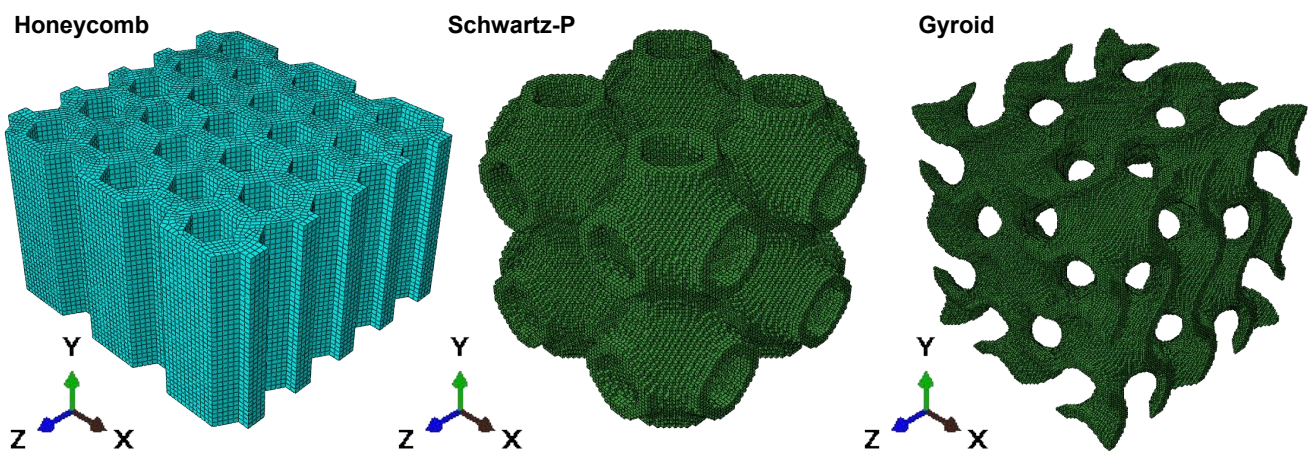
822 and an initial increment of 0.01 was applied. Because of the complexity of the model, the minimum

823 increment allowed was set to 1.0×10^{-9} . In the assembly section, the two planes were positioned on

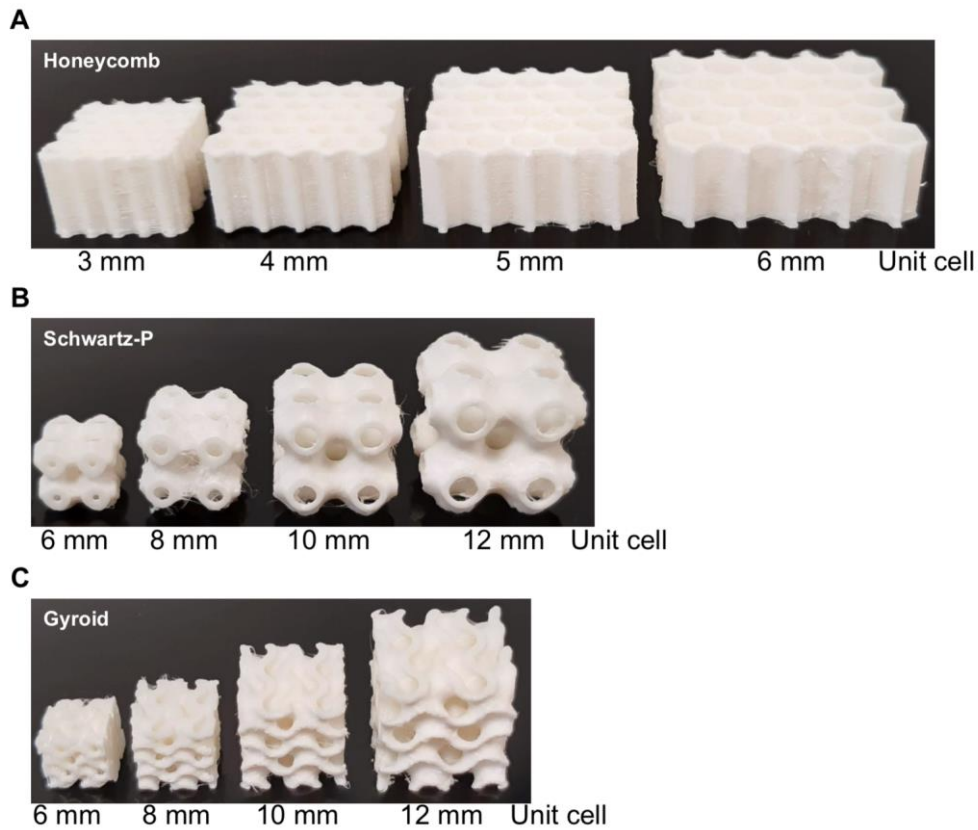
824 the top and the bottom of the model and constrained with the model using a TIE type constraint.

825 Then, two different types of interactions were defined. One type considers the frictionless contact

826 between the model and the two planes. The second one considers the self-contact among the
827 surfaces in the honeycomb model to avoid auto intersection once compressed.
828 Then, the meshing procedure was conducted, selecting C3D10 quadratic tetrahedral elements with
829 an average size of 0.7 mm. For the hyperelastic simulation, the element selected was C3D10H,
830 where H stands for the hybrid formulation, enabling calculation with incompressible materials (i.e.,
831 $D = 0$). The two planes representative of the fixed and moving portion of the compressing machine
832 were meshed with R3D4 quadrilateral elements, with an average mesh size of 4 mm (there was no
833 need to use a finer mesh).
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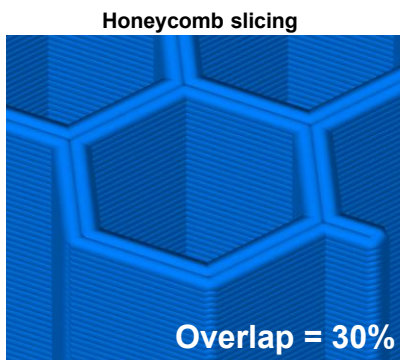
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838 **Figure S1.** Meshed models of honeycomb, schwartz-P, and gyroid structures used for the simulation data.
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Figure S2. Tunable 3D printed soft lattice scaffolds (A) honeycomb, (B) schwartz-P, and (C) gyroid with optimized printing parameters.

Deposition pathway



Number of Honeycomb samples	Measurements of 3D printed wall [mm]
1	1,030
2	1,086
3	1,213
4	1,081
5	1,104
6	1,010
7	1,103
8	1,003
9	1,053
10	1,029
Mean [mm]	1,071
SD	0,062

$$\% \text{ Overlap} = (2 \times \text{Path width [mm]} - \text{Design thickness [mm]}) / \text{Design thickness [mm]}$$

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Figure S3. Deposition pathway, percentage of overlap and the thickness of the 3D printed walls.

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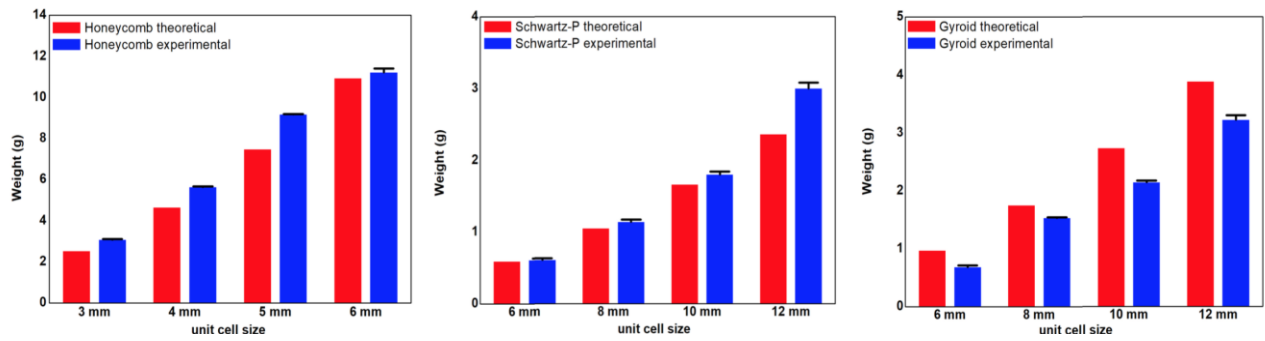
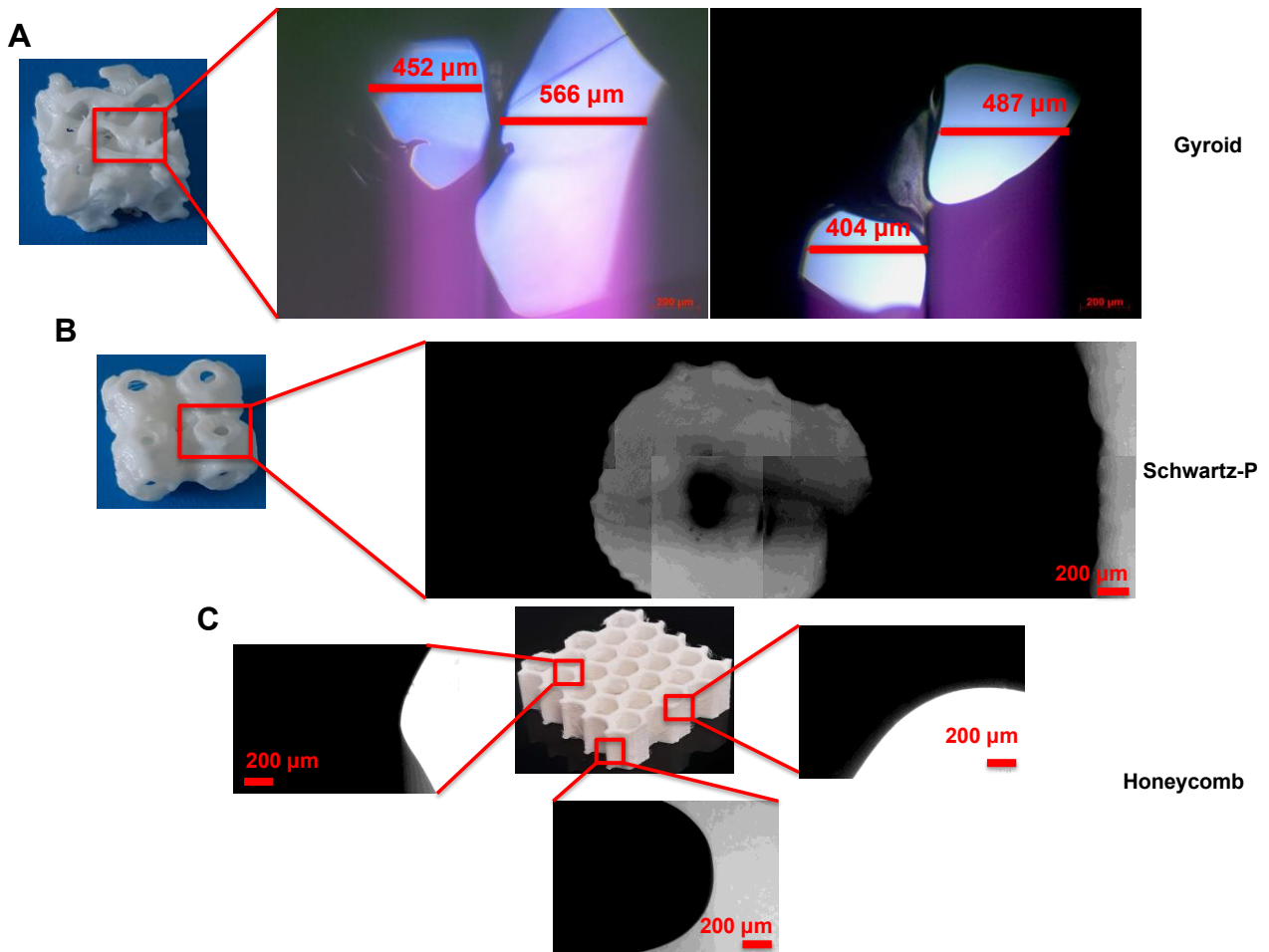
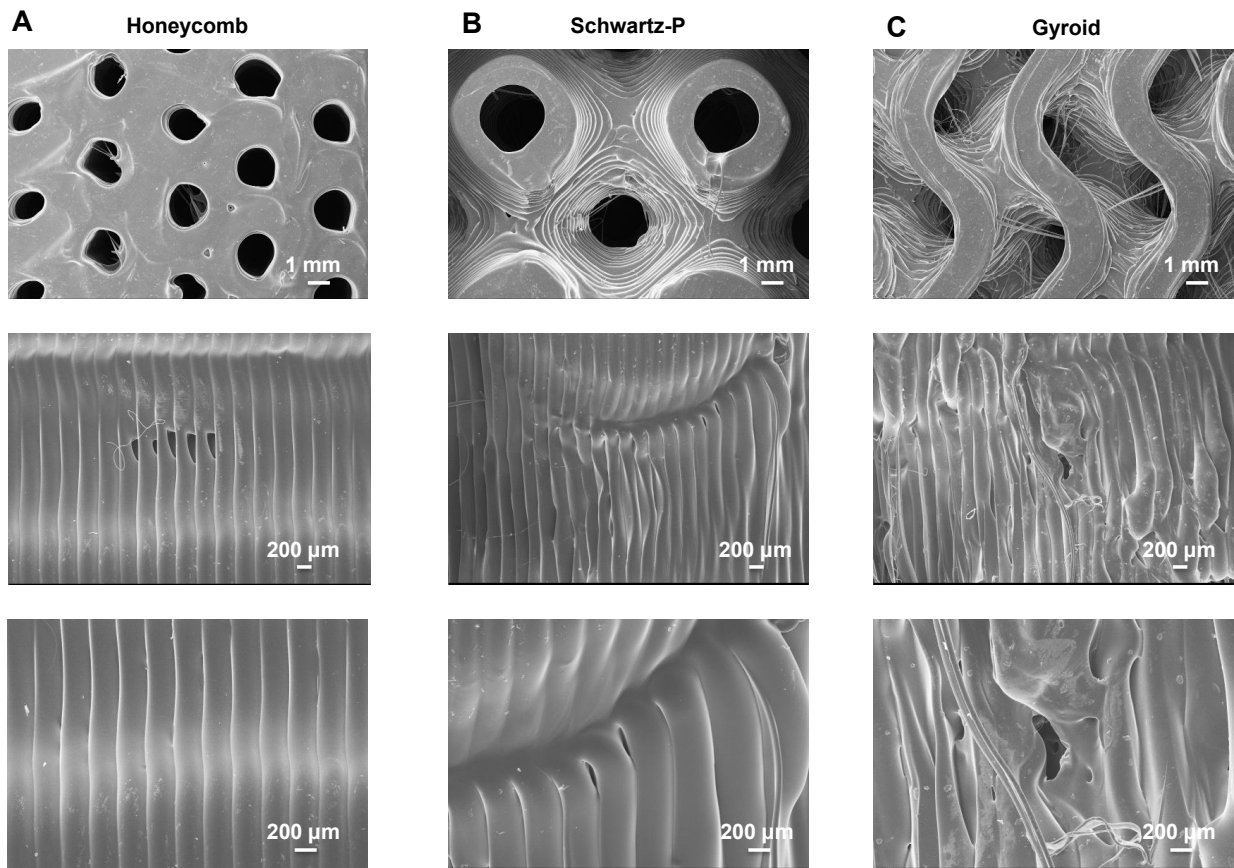


Figure S4. Comparison between the theoretical and experimental weight of 3D printed lattice structures with different geometry and unit cell size.



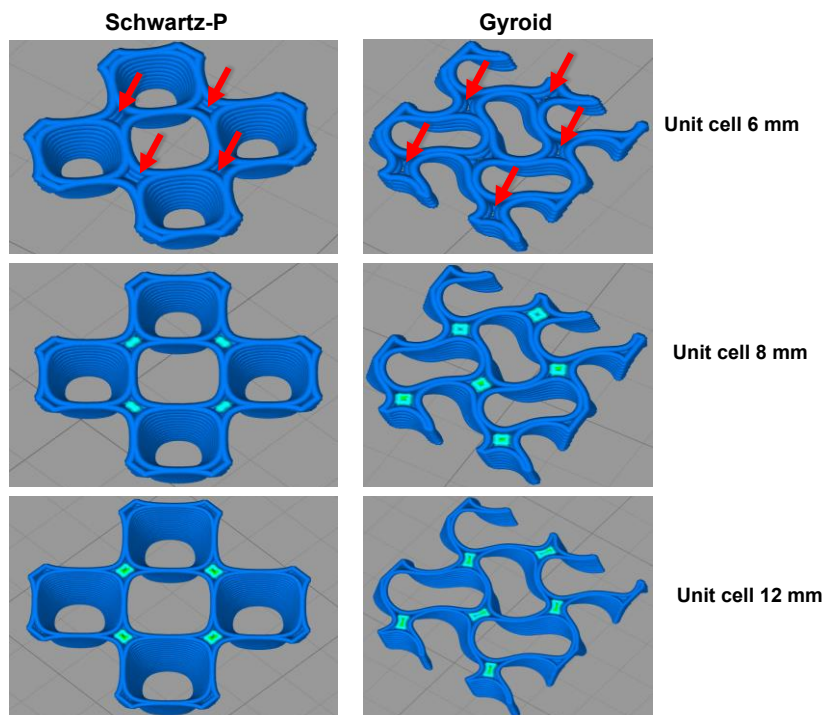
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Figure S5. Optical microscopy images of scaffolds cross-section showing (A) micrometric cavitations within the gyroid structures; (B-C) no micrometric cavitations were identified in the schwartz-P and honeycomb structures.



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Figure S6. SEM images of (A) honeycomb, (B) schwartz-P, and (C) gyroid structures. No defects such as sagging, improper layer adhesion or porosity are observed in the honeycomb, and schwartz-P structures; whereas microscopic pores are instead present in the gyroid structures.



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Figure S7. Layers deposition during bridging of TPMS structures with 6, 8, and 12 mm of unit cell size. Red arrows indicate the presence of voids during layer deposition visible only in structures with a unit cell size of 6 mm.

Printing Parameters	H3	H4	H5	H6	SP6	SP8	SP10	SP12	G6	G8	G10	G12
Extruder diameter [mm]	0.4	0.4	0.4	0.4	0.4	0.4	0.4	0.4	0.4	0.4	0.4	0.4
Extruder width [mm]	0.65	0.65	0.65	0.65	0.5	0.5	0.5	0.5	0.5	0.5	0.5	0.5
Extrusion multiplier [-]	1	1	1	1	1	1.1	1.1	1.1	1	1	1	1
Retraction distance [mm]	1	0	0	0	1	1	1	1	1	1	1	1
Retraction speed [mm/min]	2,400	0	0	0	3,000	3,000	3,000	3,000	3,000	3,000	3,000	3,000
Wipe distance [mm]	3	3	3	3	2	2	2	2	0.5	0.5	0.5	0.5
Layer height [mm]	0.15	0.15	0.15	0.15	0.15	0.15	0.15	0.15	0.15	0.15	0.15	0.15
Infill [%]	100	100	100	100	100	100	100	100	100	100	100	100
Support	NO	NO	NO	NO	NO	NO	NO	NO	NO	NO	NO	NO
Nozzle Temperature [°C]	230	230	230	230	230	230	230	230	230	230	230	230
Bed Temperature [°C]	50	50	50	50	50	50	50	50	50	50	50	50
Fan speed [%]	0	0	0	0	20	20	20	20	10	10	10	10
Printing speed [mm/min]	1,100	3,000	3,000	3,000	1,000	2,400	2,400	2,400	1,000	2,400	2,400	2,400
Outline underspeed [%]	70	70	70	70	70	70	70	70	70	70	70	70
Solid underspeed [%]	50	50	50	50	50	50	50	50	50	50	50	50

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Table S1. Optimized printing parameters for honeycomb (H), schwartz-P (SP), and gyroid (G) with different unit cell sizes and equal thickness of 1 mm.

Honeycomb				Schwartz-P				Gyroid			
Unit cell size (mm)	E _x (MPa)	E _y (MPa)	E _z (MPa)	Unit cell size (mm)	E _x (MPa)	E _y (MPa)	E _z (MPa)	Unit cell size (mm)	E _x (MPa)	E _y (MPa)	E _z (MPa)
3	6.00	5.69	14.79	6	1.57	1.57	1.57	6	4.56	4.51	4.62
4	3.32	3.22	11.42	8	0.79	0.79	0.79	8	2.90	2.86	2.87
5	1.94	1.92	9.05	10	0.61	0.61	0.61	10	2.10	2.05	2.06
6	1.28	1.29	6.28	12	0.44	0.44	0.44	12	1.56	1.52	1.56

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Table S2. Comparison of compressive elastic moduli in the x , y , and z directions obtained from the Mooney-Rivlin hyperelastic model of honeycomb, schwartz-P, and gyroid matrix soft lattice scaffolds with different unit cell sizes.